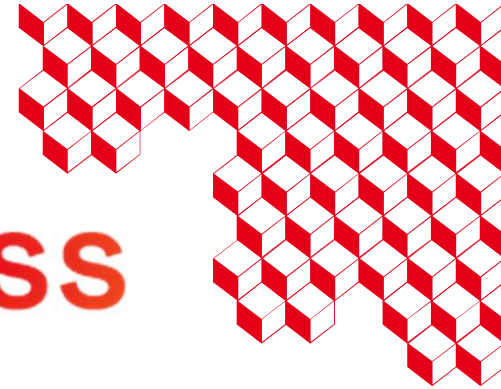




isds

sumglass



Thermodynamic models for phase separation and crystallization in nuclear glasses

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Context: The origin of some poorly soluble phases

Current nuclear French glass R7/T7 is homogeneous at its synthesis temperature...

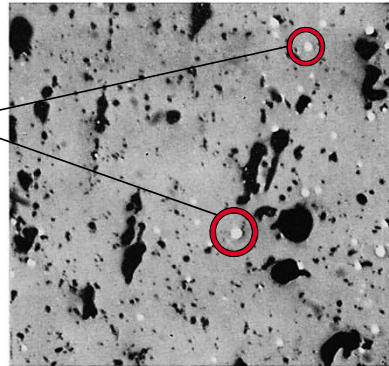
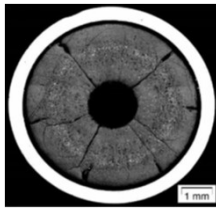
But, in a view to increasing waste loading, typical insoluble precipitates can form such as:

1. Platinum Group Metal (PGM),
2. Complex molybdates

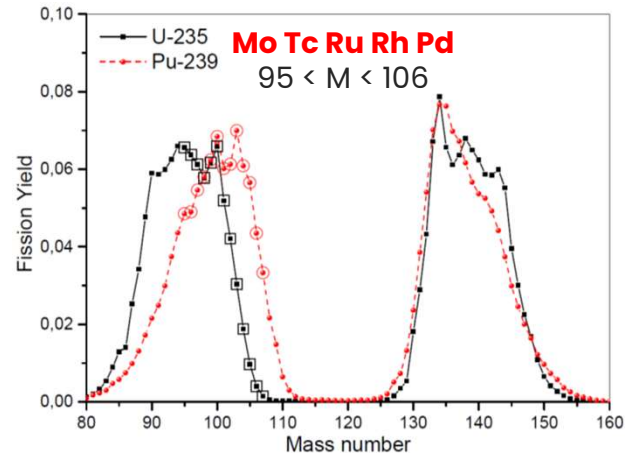
These two types of phases come mainly from the “white phase” or “noble metal phase”

This alloy is made of Pd-Rh-Ru-Mo-Tc fission products

White phases: Pd-Rh-Ru-Mo-Tc alloys
in irradiated $(U_{0,85}Pu_{0,15})O_2$ fuel [1]



[1] Bramman & al. ; J. of Nuclear Materials (1968)

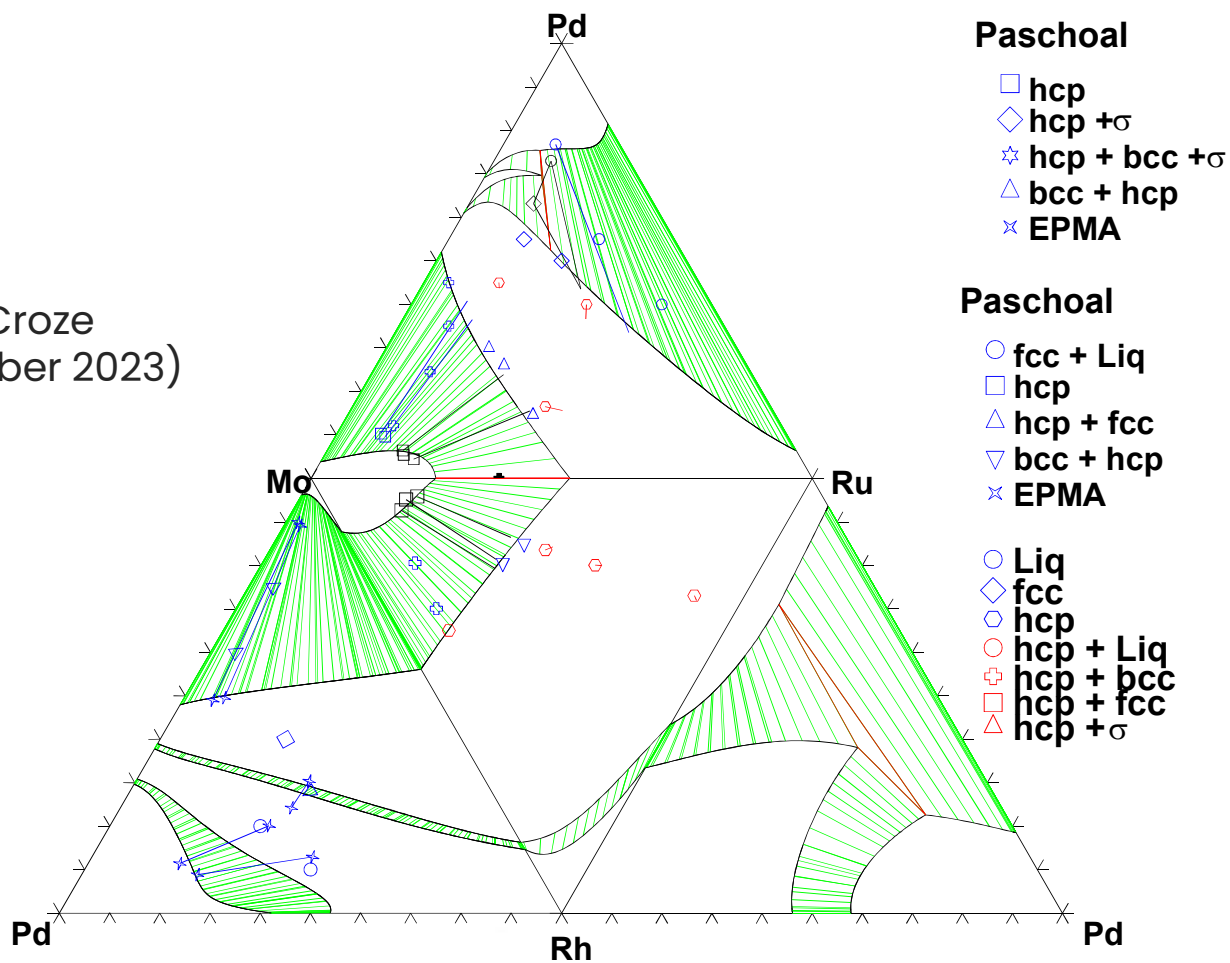


This study focuses on the fundamental research conducted to describe the thermodynamics of these complex phases formed from the noble metal phase: PGM alloys and molybdates

Context: Thermodynamics of "white phases"

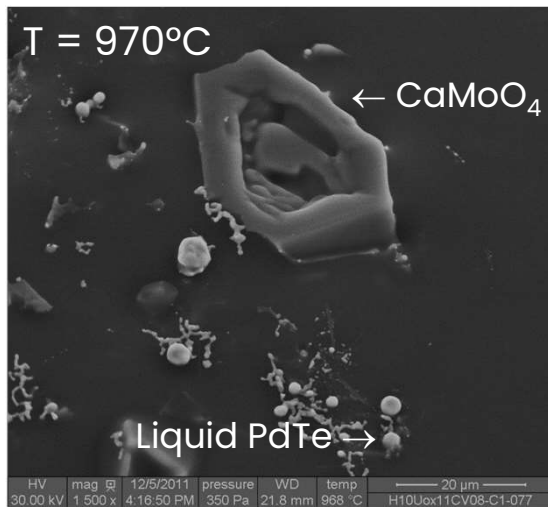


Ph.D. work of Théo Croze
(Defense in December 2023)



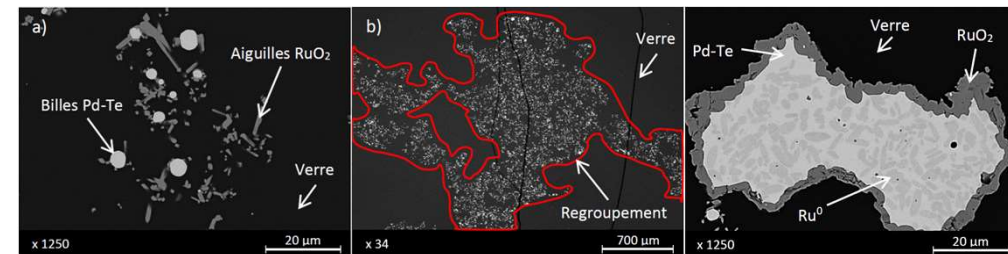
Objectives: acquisition and modelling of fundamental thermodynamic data

The objective of this work is to predict the formation and morphology of typical insoluble phases in prototypic nuclear glasses such as PGM precipitates and complex molybdates that can form beyond their solubility limits



CaMoO₄ observed by ESEM in situ in during UOX glass synthesis at 970°C

In the UOX glass without Rh, the metallic liquid containing mainly Pd-Te is liquid at 960 °C



Backscattered SEM images from C. Laurin PhD, 2019
a) typical shape of RuO₂ and Pd-Te particles,
b) a cluster of insoluble particles in an unstirred glass
c) Pd-Te-Ru cluster.

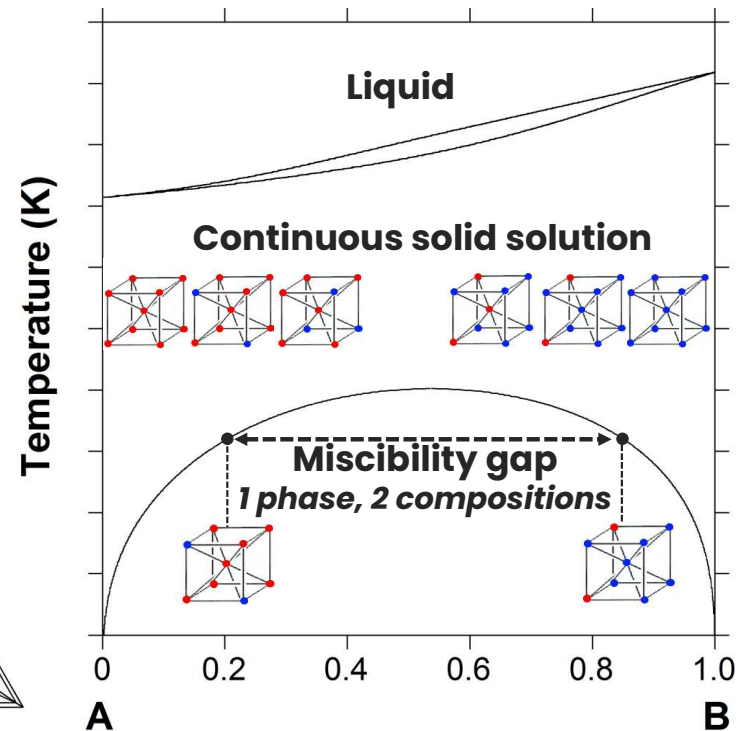
Thermodynamic origin of phase separation: Liquid miscibility gap

A single phase (liquid or solid) in which **TWO** distinct composition sets cannot mix: **ONE** phase, **ONE** structure, **ONE** Gibbs energy function but **TWO** compositions

Liquid miscibility gap

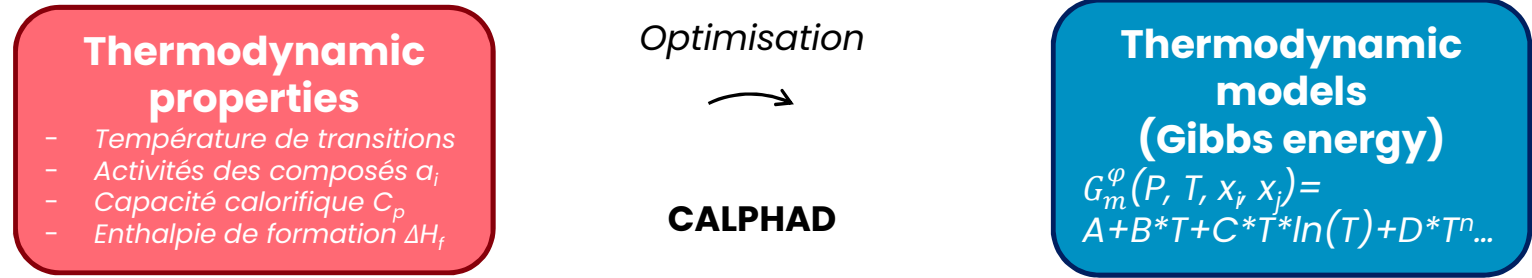


Solid miscibility gap





Approach: Thermodynamic modelling by the CalPhad method (Calculation of Phase Diagram)



- Thermodynamic properties**
- *Température de transitions*
 - *Activités des composés a_i*
 - *Capacité calorifique C_p*
 - *Enthalpie de formation ΔH_f*

Thermodynamic models (Gibbs energy)

$$G_m^\phi(P, T, x_i, x_j) = A + B \cdot T + C \cdot T \cdot \ln(T) + D \cdot T^n \dots$$

- Available data**
Experiments & Atomistic calculations
- DTA
 - DSC
 - High temperature pyrometry
 - XRD
 - SEM-EDS
 - ICP-MS
 - DFT + MD

1. Critical analysis of literature data
2. If lack of data, new experiments are performed
3. Also atomistic – DFT – calculations

- Pd-Rh-Ru-Te:
- Heat treatments
 - DTA

Calphad is the only predictive method for complex systems under all conditions: P, T, x_i
The developed database is used to perform application calculations: Phase diagram, thermodynamic properties

Some relevant data on PGM & chalcogen phases

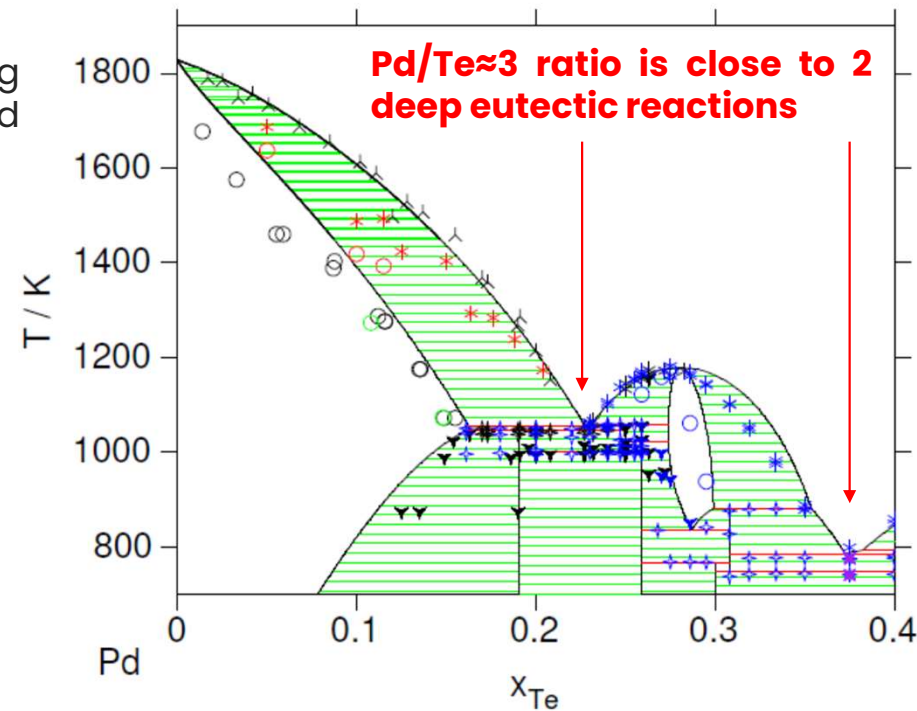
The Pd-Te phase diagrams

The formation of Pd-Te precipitates is linked to the strong interaction between Pd and Te (No Te \rightarrow no metallic liquid droplets)

Pd-Te alloys are characterized by their relatively low melting points:

- 1st reaction: @ 1053 K & $x_{Te}=0.228$ in at. fr.
- 2nd reaction: @ 768 K & $x_{Te}=0.375$ in at. fr.

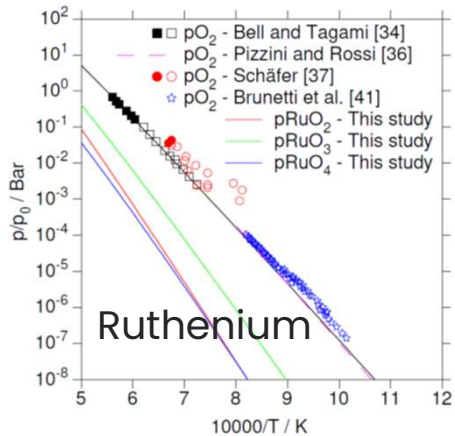
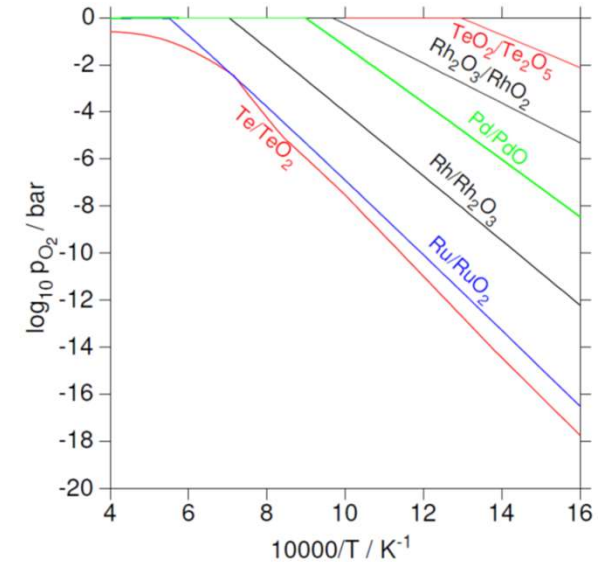
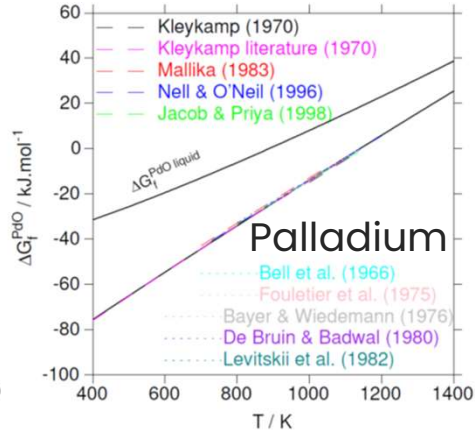
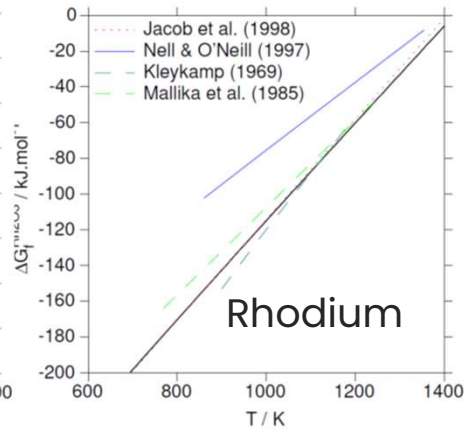
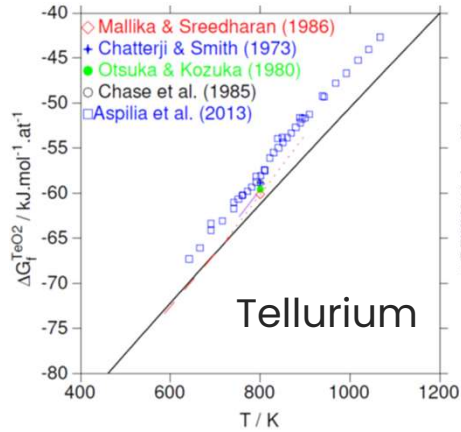
The Pd-Te binary system was the first one to be published in the framework of PGMs interactions in HL nuclear waste





Some relevant data on PGM & chalcogen oxide phases

Thermodynamic properties: Gibbs energies of oxide phases



Controlling the RedOx of the elements involved in the interactions at the origin of precipitation phenomena is a key point in this thermodynamic approach. In the specific case of PGMs, it enables us to predict speciation between the metallic or oxide phases.

Calculations of PGMs thermodynamic properties

Focus on Ruthenium speciation



Because PGMs are poorly soluble, their thermodynamic properties can be calculated without considering the glass chemistry → The interaction with the melt is only taken into account through its oxygen potential

Case of an UOX waste composition after reprocessing:

$\text{TeO}_2=0.3$, $\text{RuO}_2=1.3$, $\text{Rh}=0.2$, $\text{Pd}=0.64$ (in g/l in the solution)

Temperature is fixed to 1473 K, calculations are performed as a function of $p(\text{O}_2)$

Calculations of PGMs thermodynamic properties

Focus on Ruthenium speciation



Question 1: In which phases are the PGMs elements?

If $\log p(\text{O}_2) < -1.8$:

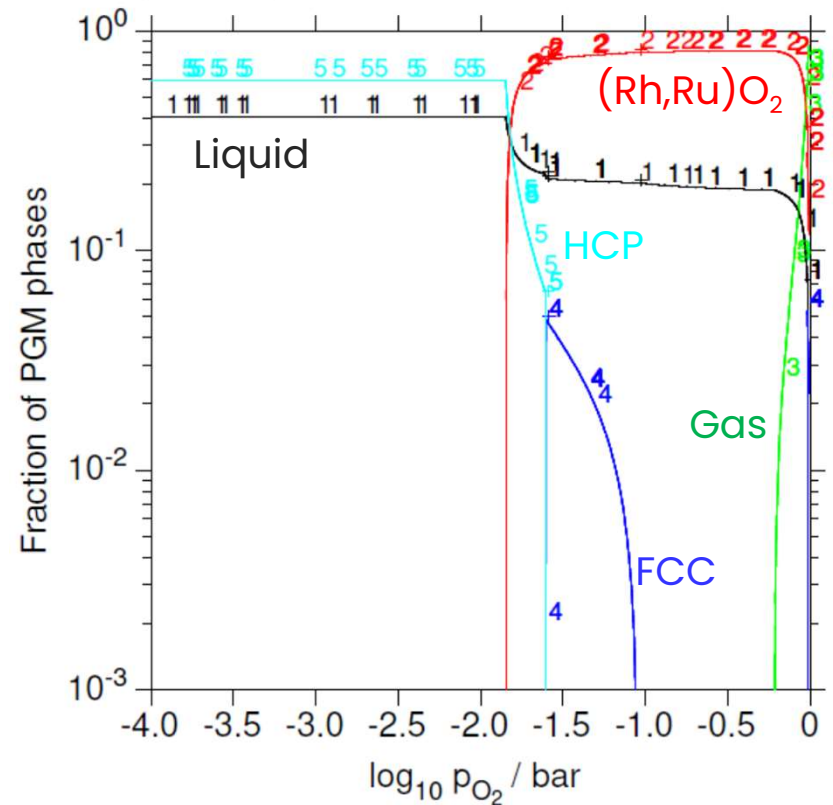
All the elements are reduced in the metallic liquid and the HCP alloy

If $-1.8 < \log p(\text{O}_2) < -1.1$:

$(\text{Rh,Ru})\text{O}_2$ forms, the amount of metallic alloys decreases rapidly

If $-1.1 < \log p(\text{O}_2)$:

$(\text{Rh,Ru})\text{O}_2$ and metallic liquid are the only phases until gas forms at high O_2 pressure



Calculations of PGMs thermodynamic properties

Focus on Ruthenium speciation



Question 2: What is ruthenium speciation?

If $\log p(\text{O}_2) < -1.8$:

Nearly all the Ru is alloyed in the HCP solid solution, about 1% in the metallic liquid

If $-1.8 < \log p(\text{O}_2) < -1.1$:

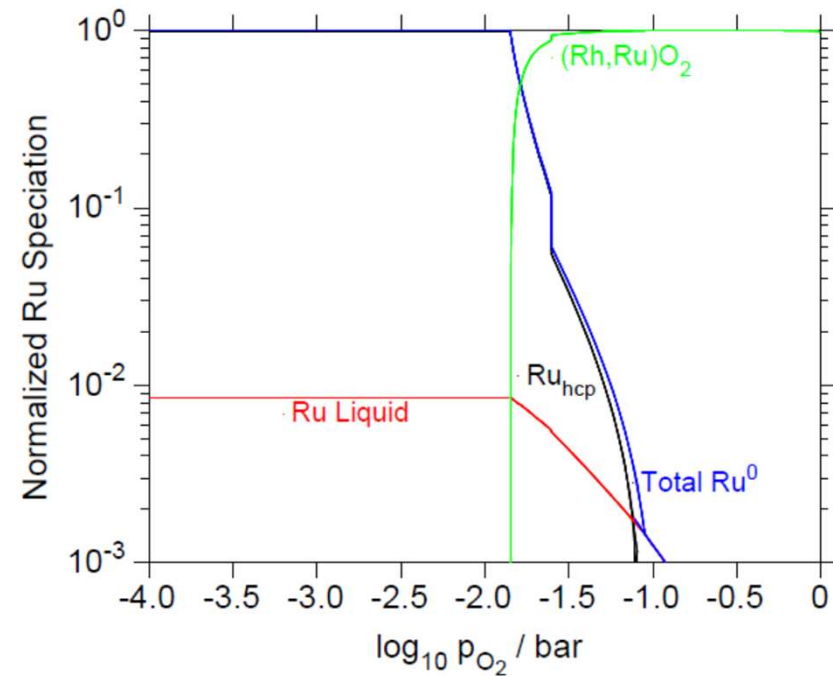
$(\text{Rh,Ru})\text{O}_2$ forms, the amount of Ru^{IV} skyrockets, the amount of alloys (HCP+FCC) decreases rapidly

If $-1.1 < \log p(\text{O}_2)$:

Ru is oxidized in $(\text{Rh,Ru})\text{O}_2$ rutile phase
Only a few amount of Ru remains in the metallic liquid...

This limit is still badly known

(The gas phase is not represented)





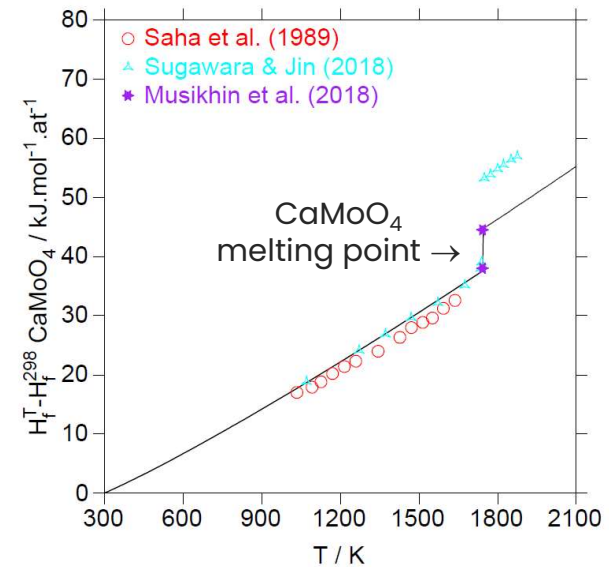
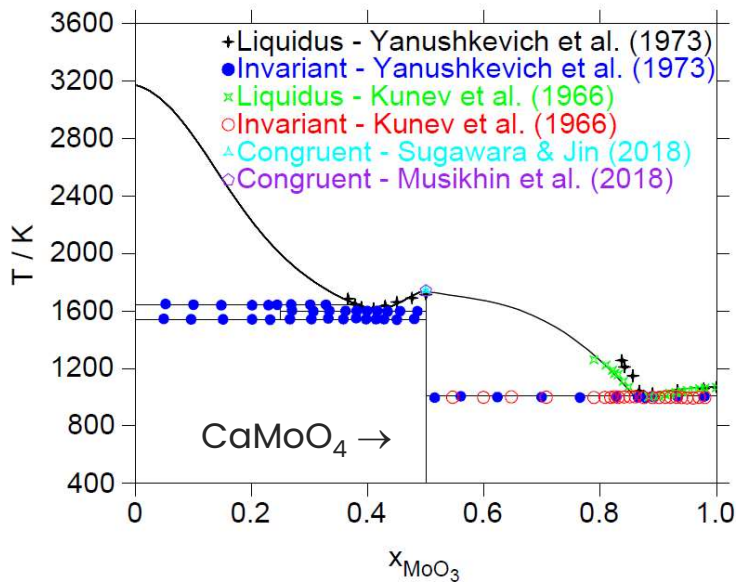
Some relevant data on calcium molybdates

Thermodynamic properties & phase diagrams

Enthalpy calculation for intermediary compounds formed during glass melting: CaMoO_4

These data can be useful to be implemented in glass melting modeling to optimize the vitrification processes (knowledge of the latent heat of fusion)

- Calculation of transition temperatures
- Prediction of heat contents



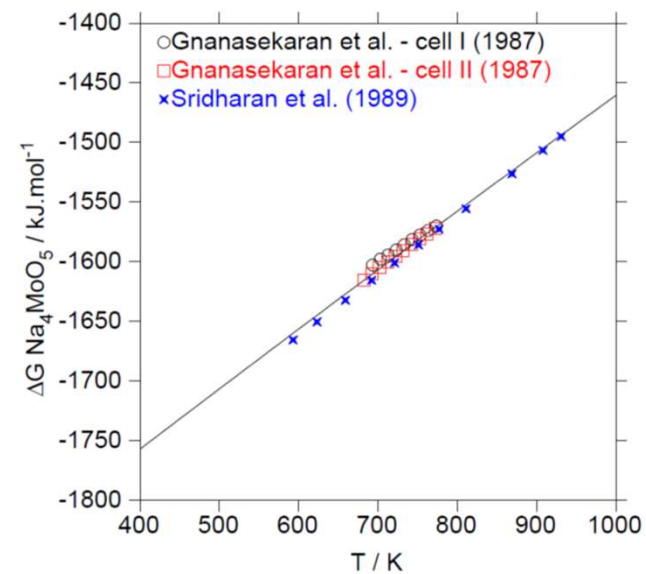
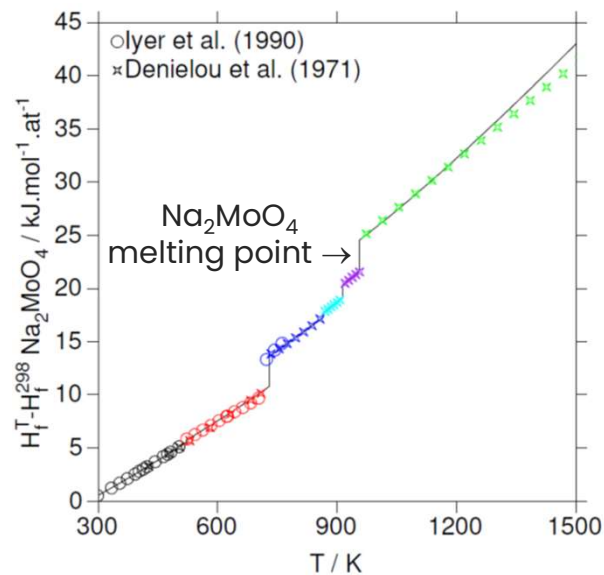
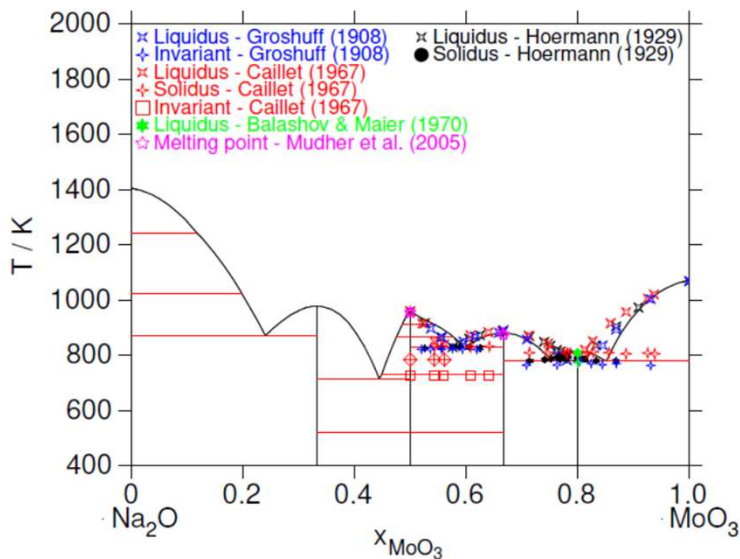


Some relevant data on sodium molybdates

Thermodynamic properties & phase diagrams

Enthalpy calculation for intermediary compounds formed during glass melting: Na_2MoO_4 allotrops
These data can be useful to be implemented in glass melting modeling to optimize the vitrification processes (knowledge of the latent heat of fusion)

- Calculation of transition temperatures
- Prediction of heat contents



Some relevant data on Na₂O-SiO₂ system

Thermodynamic properties & phase diagrams



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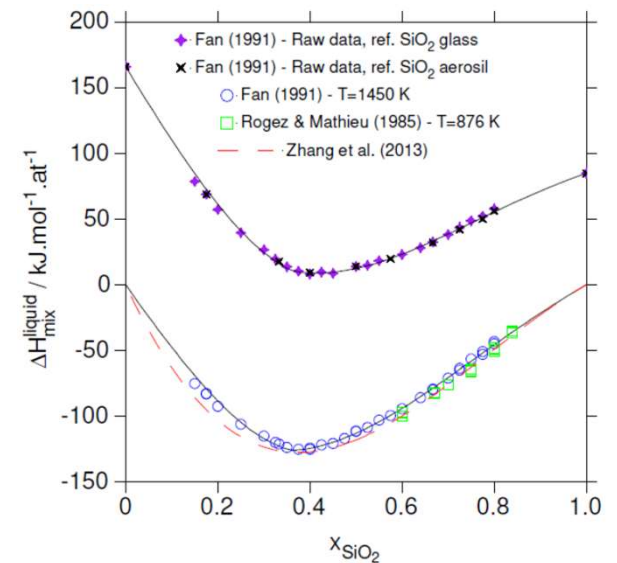
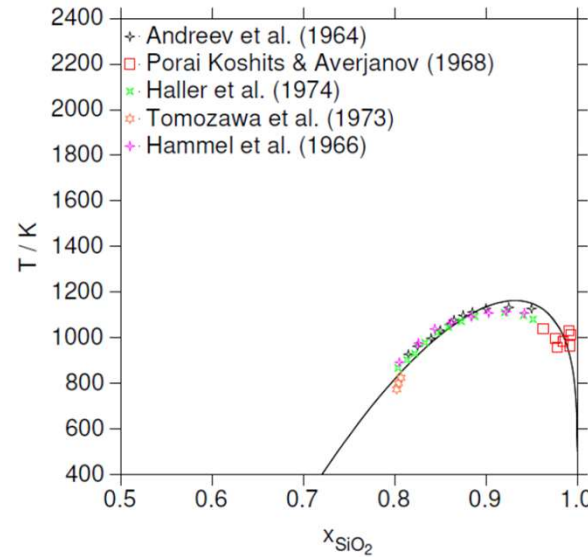
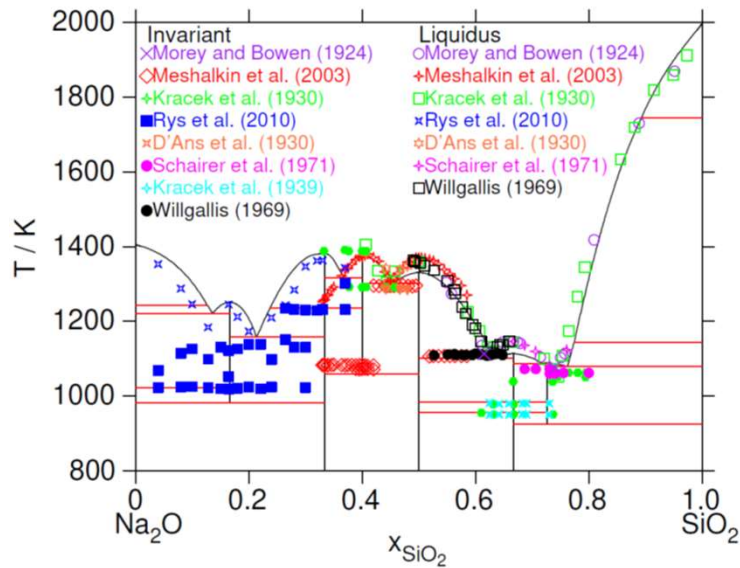


The Na₂O-SiO₂ system is a key binary for predicting glass chemistry
The model from Zhang et al. has been reassessed, with particular emphasis on modelling miscibility gaps

Thermodynamic description of the M₂O-SiO₂ (M = K, Na) systems

Ligang Zhang*, Clemens Schmetterer, Patrick J. Masset¹

Freiburg University of Mining and Technology, Centre for Innovation Competence Virtacon, Fachsmühlenweg 9, D-09596 Freiberg, Germany



The aim of this modelling is to obtain a complete description of the NBS ternary in order to couple the behavior of the insoluble phases to the thermochemistry of glass melt



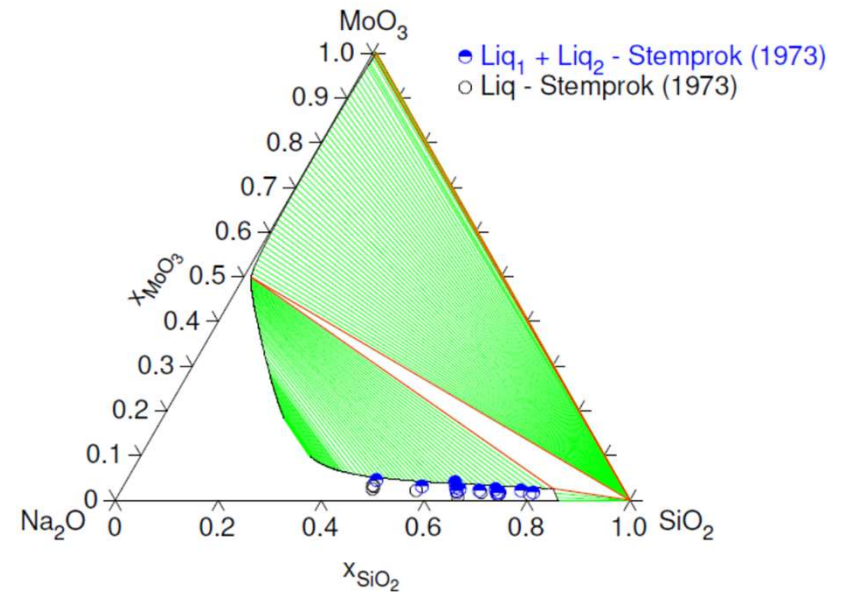
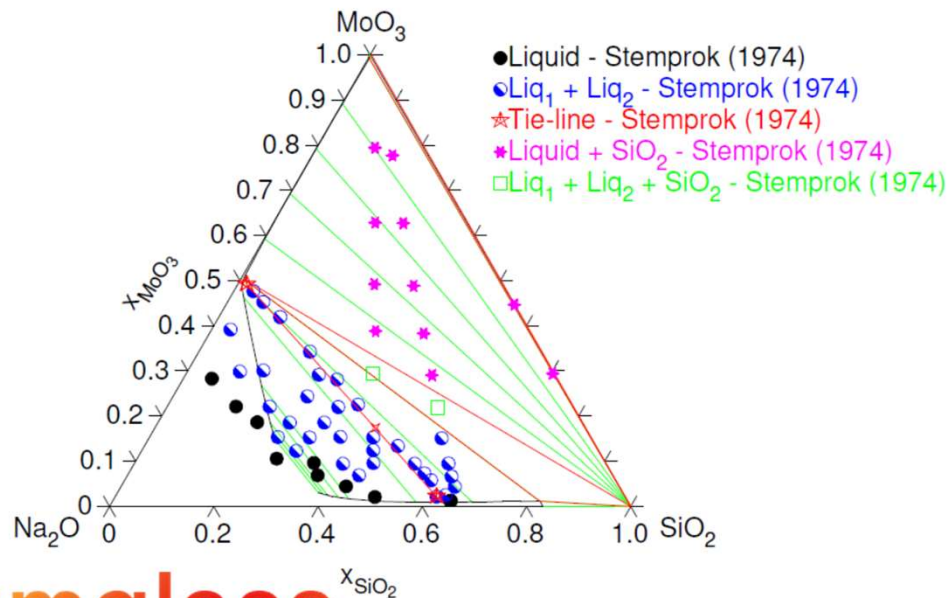
Some relevant data on sodium molybdates

Extrapolated $\text{Na}_2\text{O}-\text{MoO}_3-\text{SiO}_2$ phase diagram

The calculation of the ternary $\text{Na}_2\text{O}-\text{MoO}_3-\text{SiO}_2$ system reveals the extent of the miscibility gap between Silica rich and molybdate rich liquids

These data are used to predict the solubility limit of molybdates in the melt and the precipitation temperatures. Calculations are compared with experimental results from:

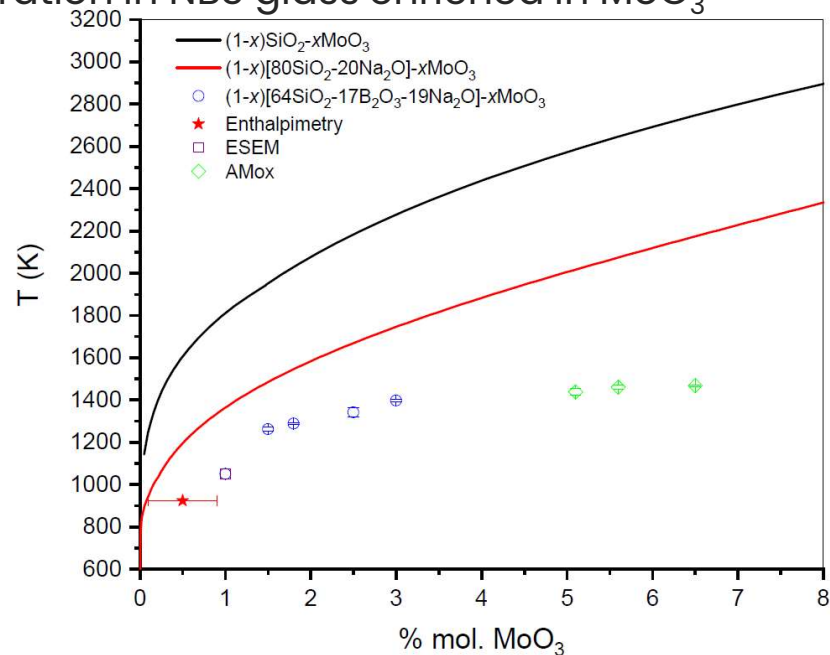
- Fig. 1: Stemprok at 1473 K
- Fig. 2: Stemprok at 1673 K



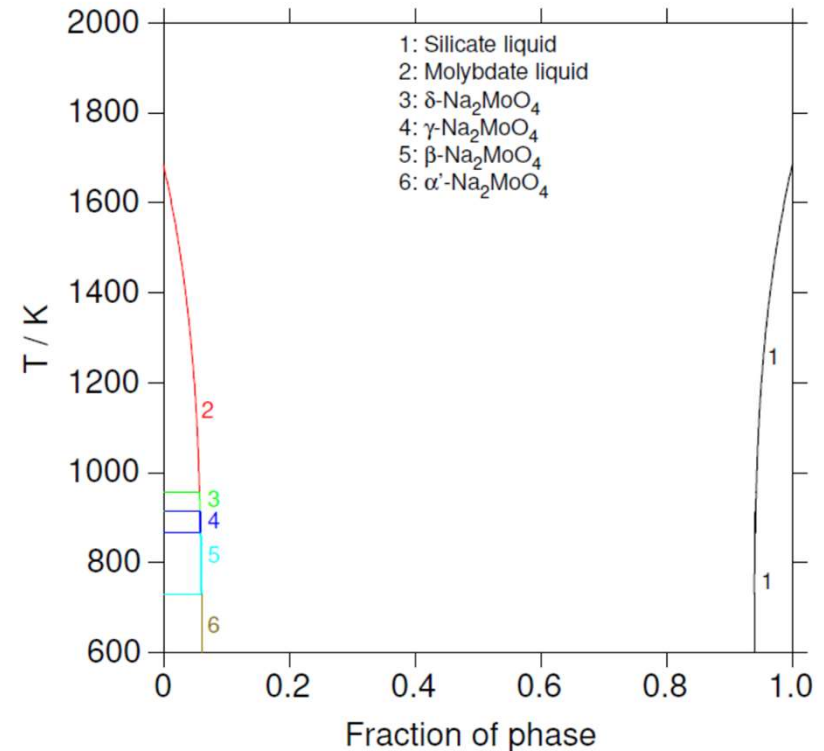
Calculation of sodium molybdate thermodynamic properties

Focus on Na_2MoO_4 formation temperature

Objective: Comparison between experimental investigations and thermodynamic calculations about liquid-liquid phase separation in NBS glass enriched in MoO_3



Phase separation temperatures as a function of MoO_3 for SiO_2 - MoO_3 & SiO_2 - Na_2O - MoO_3 . Comparison with SiO_2 - Na_2O - B_2O_3 - MoO_3 glasses and in previous study on complex AMox glass compositions



Calculated solidification path of a MoO_3 - Na_2O - SiO_2 glass having a ratio $\text{SiO}_2/\text{Na}_2\text{O} \approx 4$ and containing 3 mol.% MoO_3 .



Complementary of multiscale approaches of phase separation

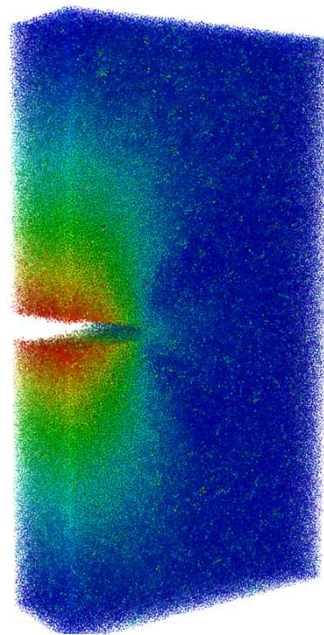
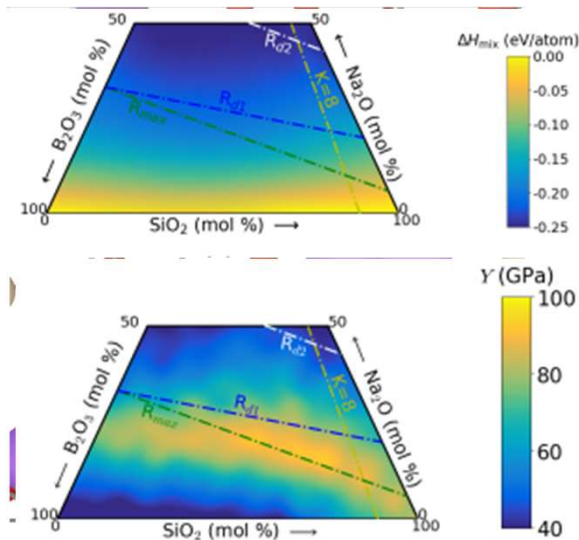
From molecular dynamics to phase-field modeling

The Calphad method complements atomistic approaches, enabling atomistic calculations to be validated and providing input data for diffusion/precipitation calculations.

Molecular dynamics

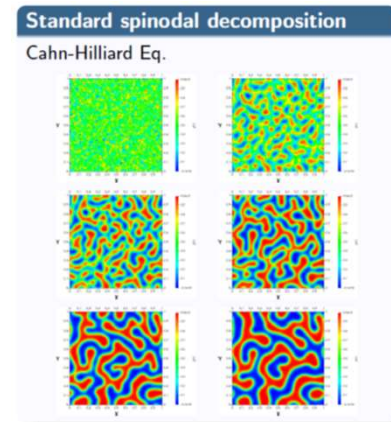
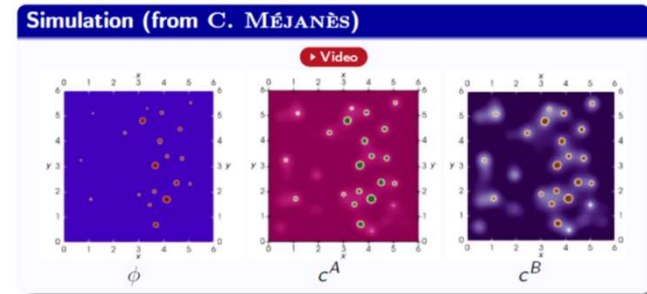
Phase-field simulations

Mixing enthalpy & Young modulus of SBNO glass
(M. Mama Toulou *et al.*, 2023)



Fissure in SBNO glass
(Ph.D. M. Mama Toulou)

Modeling Ostwald ripening



See:
Poster of C. Mejanès

Presentation of A. Cartalade.
At 11:45



Conclusion

To optimise vitrification processes, a Calphad database is under development – since 2007 – to obtain reliable fundamental data on low-solubility elements in nuclear glass. The thermodynamic results predict the stable phases and their thermodynamic properties for all compositions and temperatures

Many thermodynamic properties of PGMs & molybdate phases can be calculated $f(T, x_i)$:

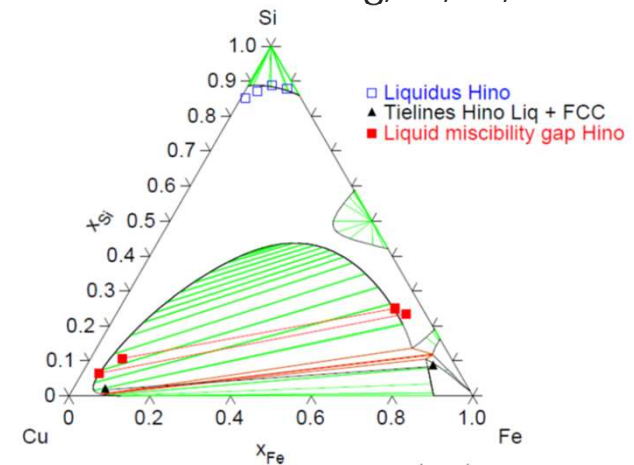
- PGMs & molybdates phase stability domains
- Redox and speciation of Pd/Rh/Ru
- Help to understand the origin of the different morphologies involving Ru⁰

The results from this database are now coupled with modelling tools dedicated to the phase separation of molybdates

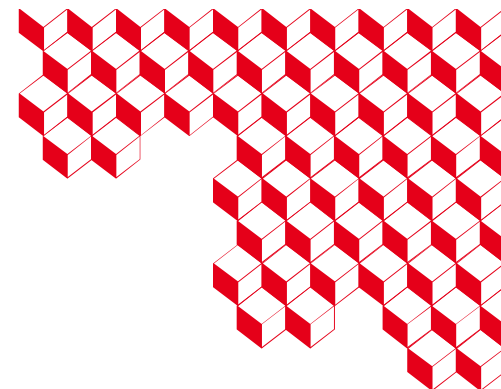
The database is still under development, NBS ternary & other elements are considered: Ag, Se, Re, Tc...

Comment

The same methodology was applied – with relevant results – to the PIVIC process to model the formation of Al-Cu-Fe-Si-U alloys in a Al₂O₃-CaO-SiO₂ matrix



02/06/2023



Merci de votre attention

Thanks to: Pierre Bénigni, Sébastien Bordier, Annabelle Laplace, Cloé Laurin, Luiz Pereira, Damien Perret, Renaud Podor, Elise Régnier

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