

# Elucidating glass (and alteration gel) structure with a combined MD and NMR approach boosted by Machine Learning

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université  
PARIS-SACLAY

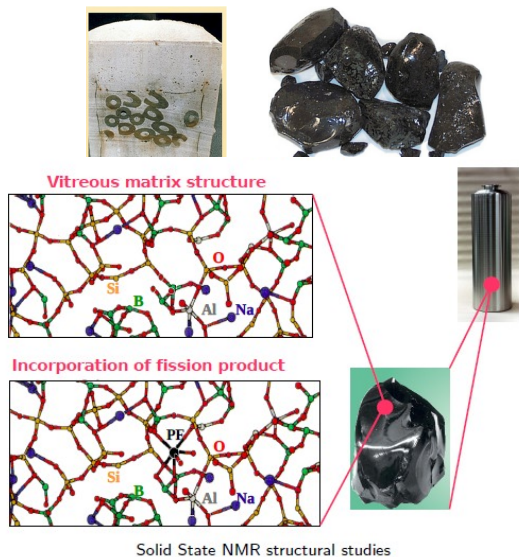
nanosciences & innovation  
nimbe



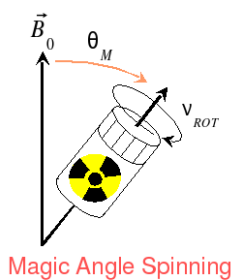
# Overview

DRF/IRAMIS/NIMBE - LSDRM - Laboratory of Structure et Dynamics by Magnetic Resonance

## Nuclear Waste Materials

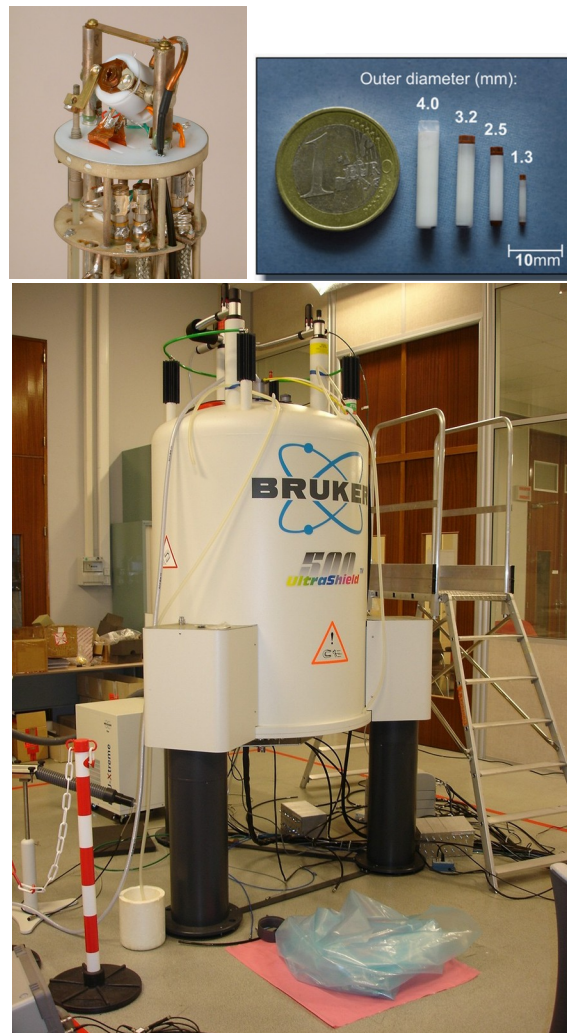


Long Term Behavior

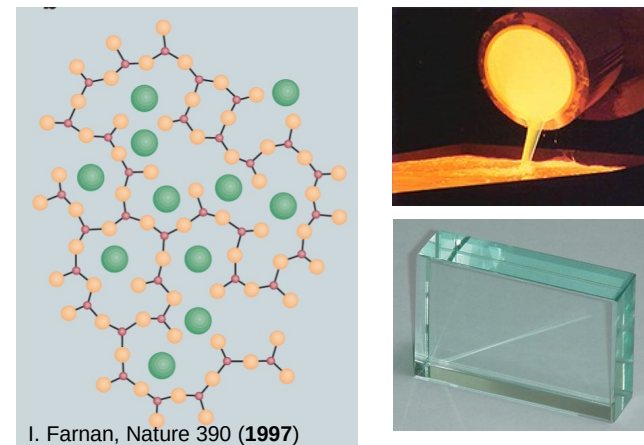


Radiological  
and  
Tritium  
MAS NMR

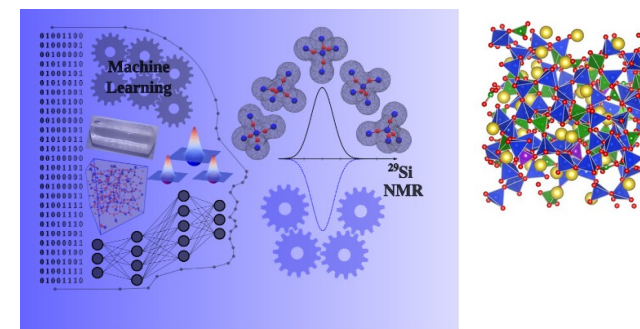
## MAS NMR & Theory



## MAS NMR of Glass & Disordered Materials



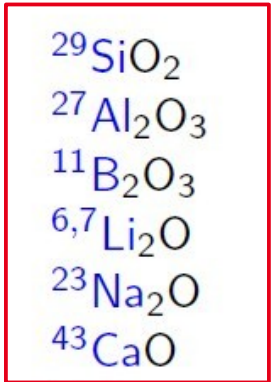
## First-Principles NMR & Machine Learning





# MAS NMR of (French) Nuclear Waste Glass

R7T7: A borosilicate glass comprising more than 30 oxides



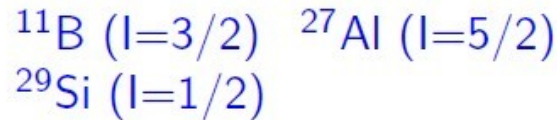
% (w)
45.12
4.87
13.92
1.97
9.78
4.01
0.99
2.48
2.89
0.28
0.41
0.50
10.35
⇒ Mo, REE (La, Nd, Ce, ...)
0.89
1.54

For solid-state NMR investigations:  
 Representative simplified (3-8 oxides)  
 compositions from the main oxides  
 contained in nuclear glass

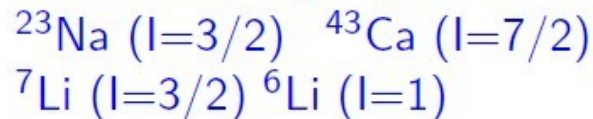


Many NMR Probes !

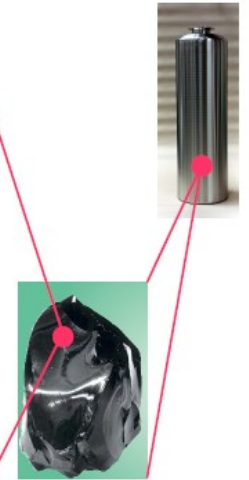
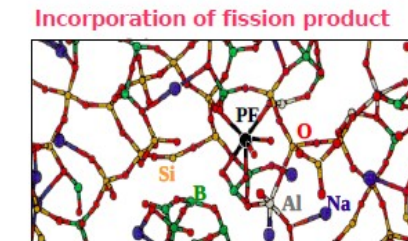
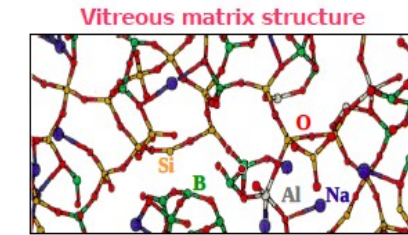
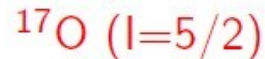
Network former cations



Network modifier / Charge compensator cations



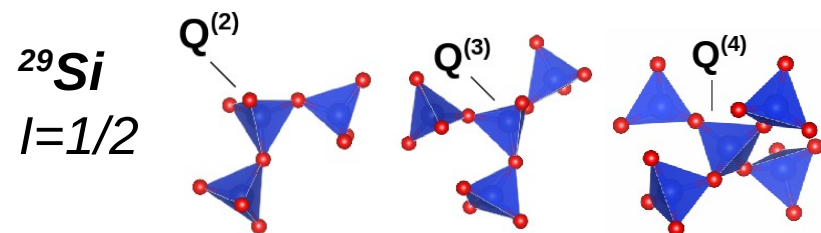
Central probe of bindings



Solid State NMR structural studies

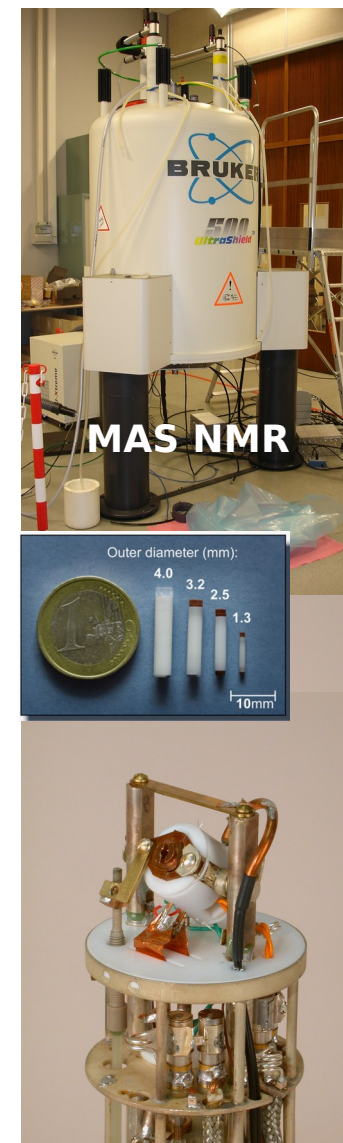
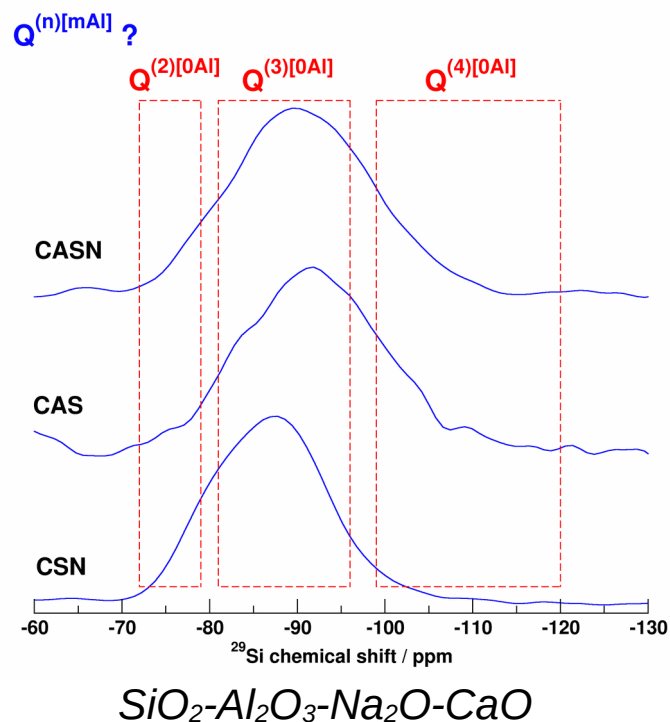
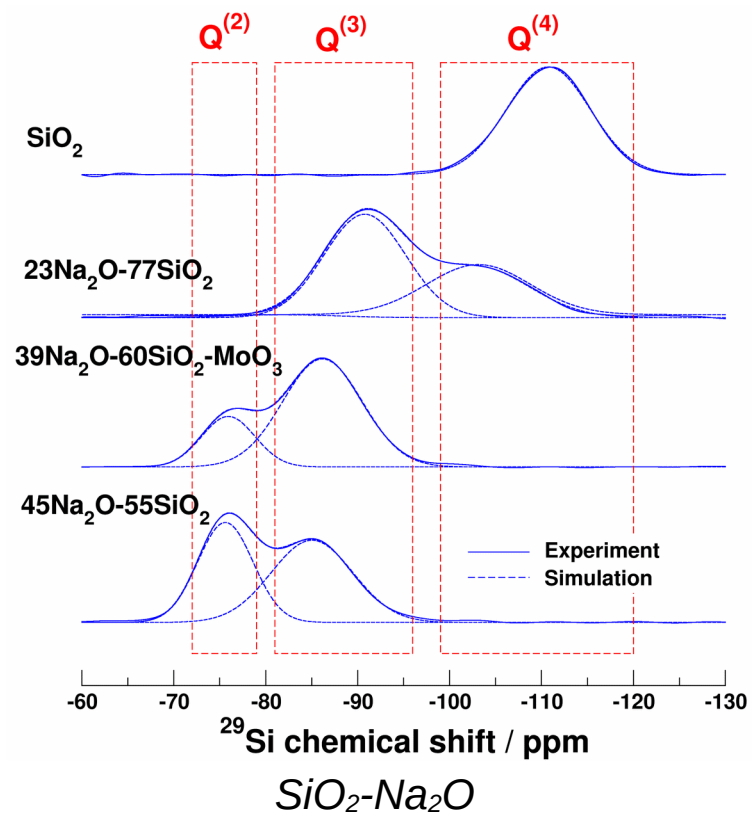
# MAS NMR in Oxide Glasses

## Deciphering the short-range order (SRO) – MAS NMR



Silicon-29 MAS NMR can resolve  $Q^{(n)}$  units in simple glass, revealing the polymerization degree.

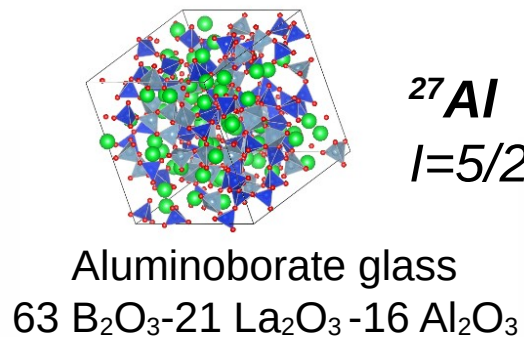
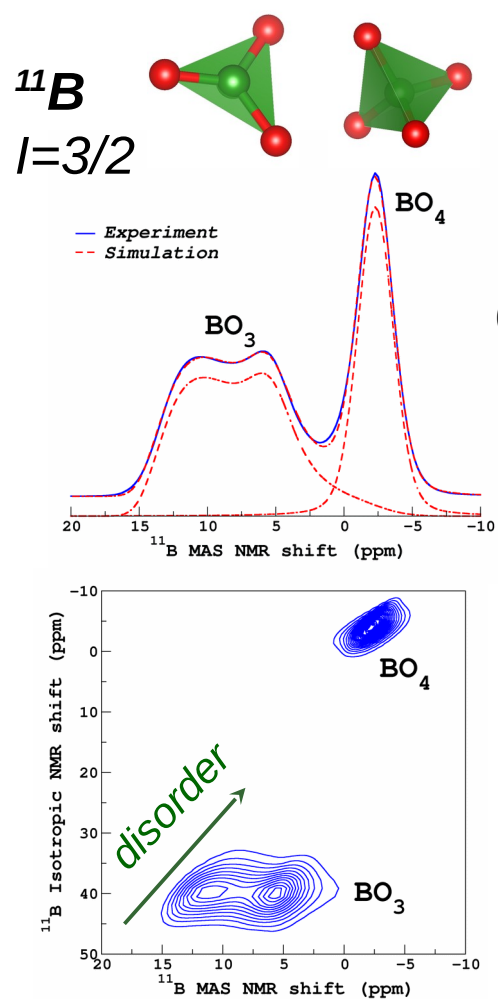
But loss of resolution for more complex composition because of Next Nearest Neighbor (NNN) impact such as  $Q^{(n)}(\text{mAl})$ ,  $Q^{(n)}(\text{mB})$ , ... Computations are the needed.



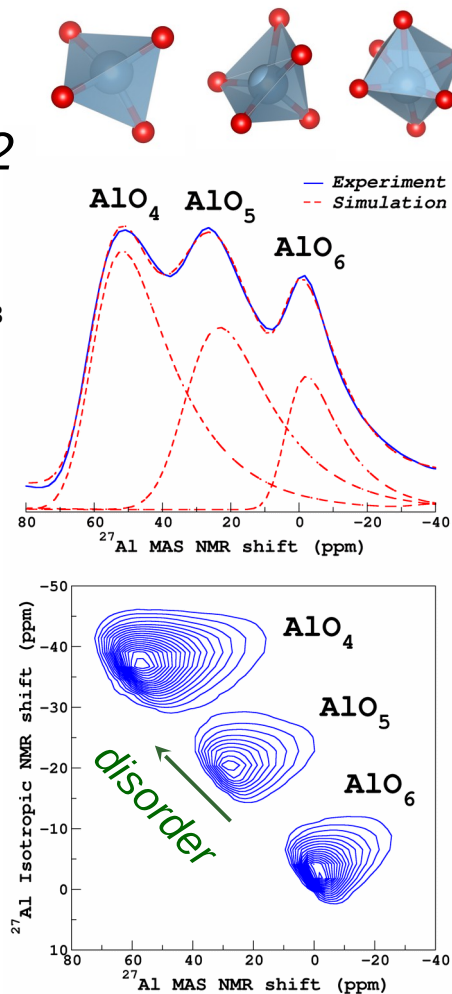


# MAS NMR in Oxide Glasses

Deciphering the short-range order (SRO) – 2D MQMAS NMR



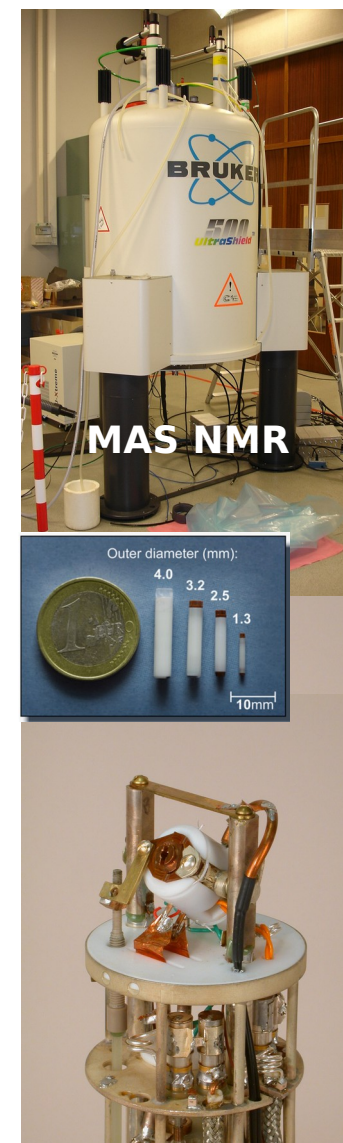
$^{27}\text{Al}$   
 $I=5/2$



1D NMR

Gain in  
resolution  
with 2D NMR

2D MQMAS NMR  
(Triple Quantum MAS)

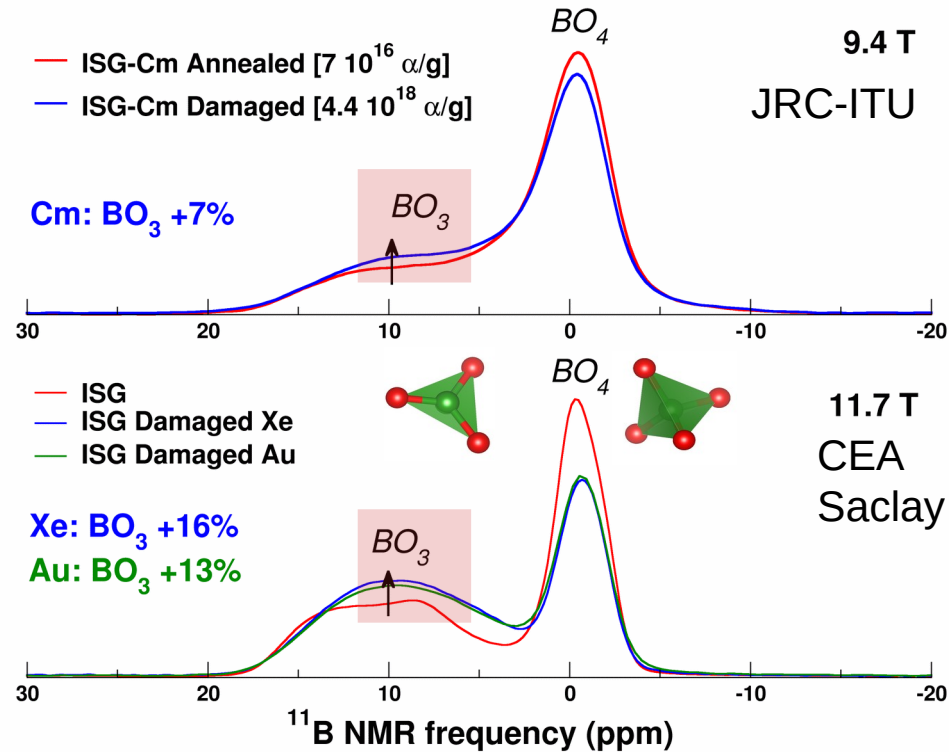


High-resolution MAS NMR reveals the structural motifs (network formers) building the glass network. NMR lineshapes are reflective of the local disorder and the underlying distribution of NMR parameters.

E. Chesneau, D Caurant, T. Charpentier,  
in preparation.

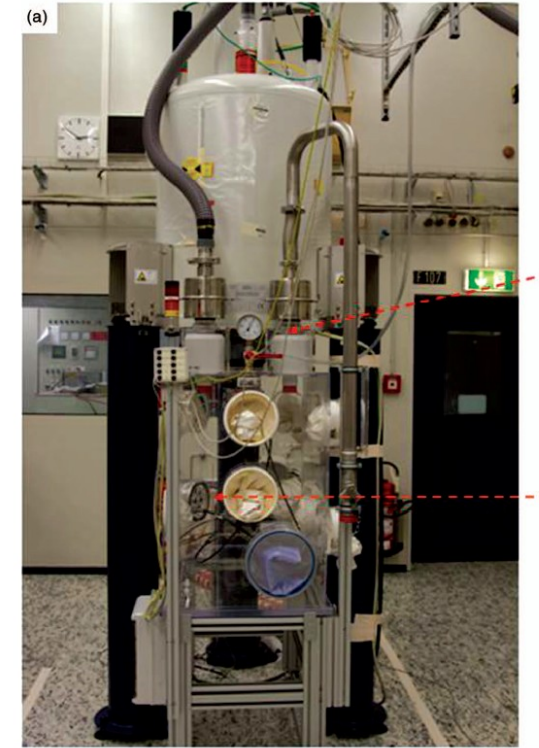
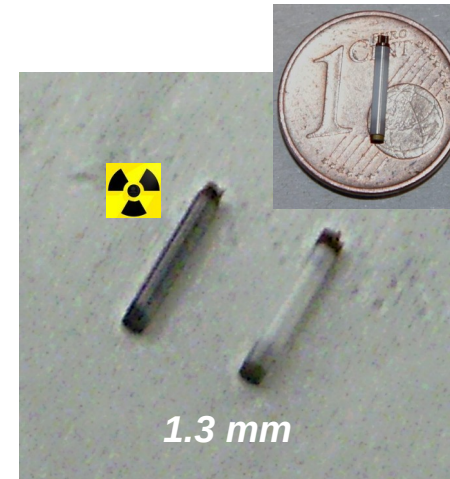
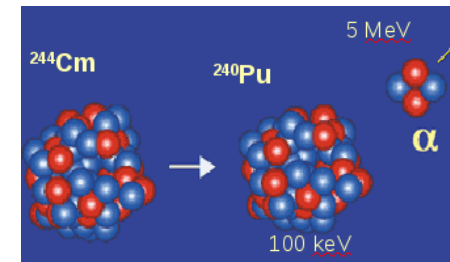
# Self-Irradiation Damages: Radiological NMR

**Radiological MAS NMR** : examination of self-irradiated glasses ( $0.1\%$  mol  $^{244}\text{Cm}$ ) and comparison with heavy ions external irradiations



$\text{BO}_4$  to  $\text{BO}_3$  conversion and other structural modifications **are lower** in alpha-decays damaged glass than in heavy ions irradiated samples (Self-Healing Effects)

**EURACT nmr**  
Trans-national Access to Unique European Actinide and Radiological NMR Facilities



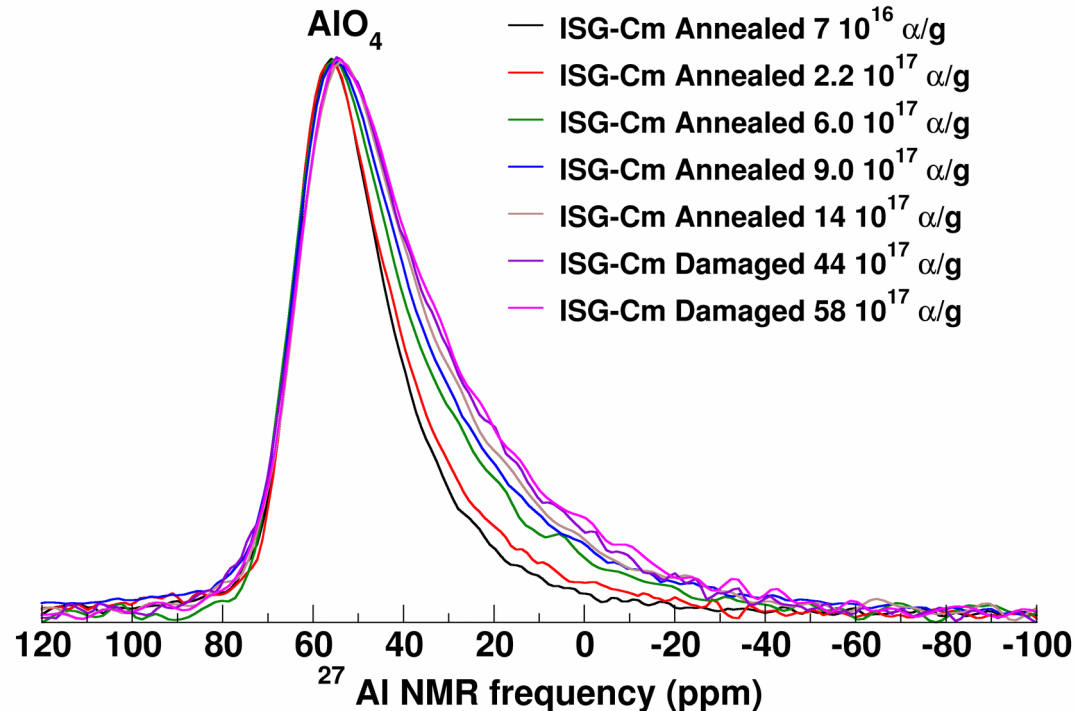
**Nuclearized NMR (9.4T) @JRC in Karlsruhe**

T. Charpentier et al., Scientific Reports 6 (2016).

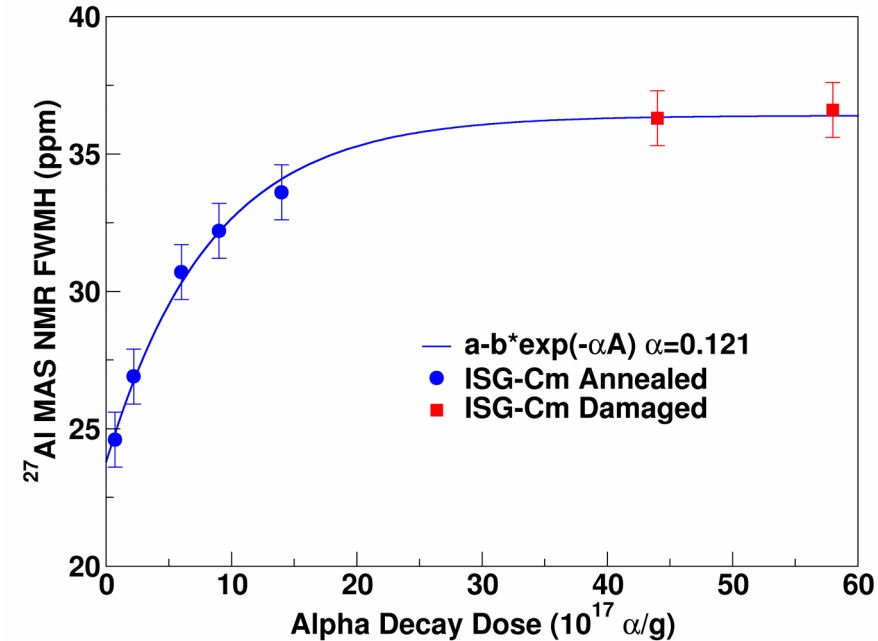
T. Charpentier, S. Peugot, EURACT-NMR Grant RP02/05 Radiation damage in nuclear glass.

# Self-Irradiation Damages : Radiological NMR

**Radiological MAS NMR** : examination of self-irradiated glasses (0.1% mol  $^{244}\text{Cm}$ )



The variations suggest a change from  $\text{AlO}_4(\text{Na}^+)$  to  $\text{AlO}_4(\text{Na}^+, \text{Ca}^{++})$ . No  $\text{AlO}_5$  is formed in contrast to external irradiation\*.

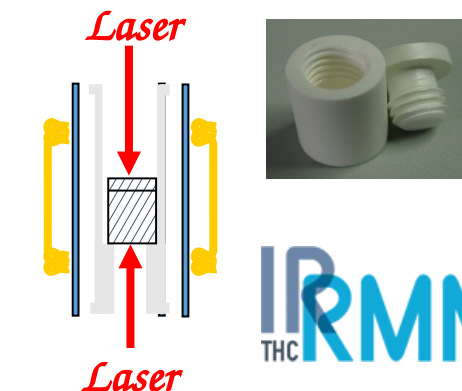
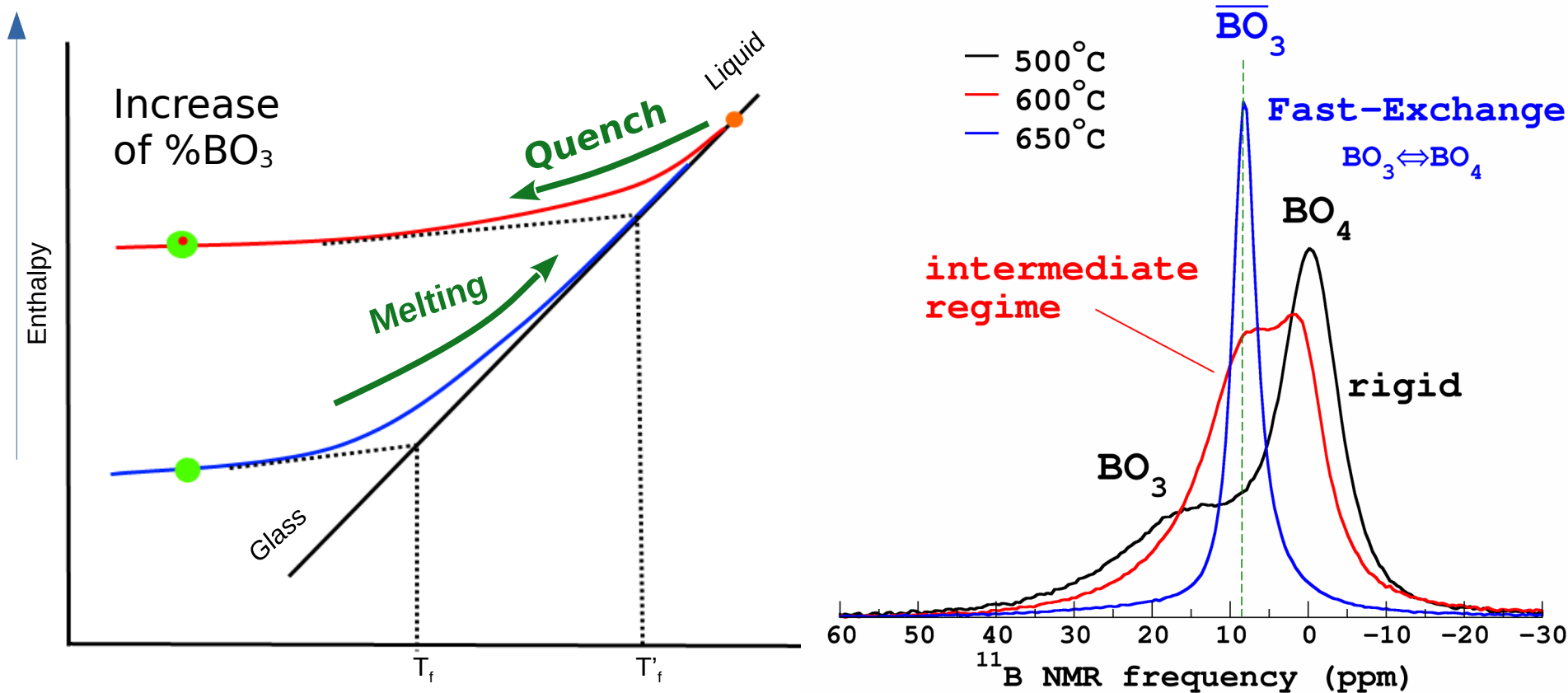


$^{27}\text{Al}$  MAS NMR evolution is well described by a *single impact model*. The damaged volume per event is estimated as  $480 \text{ nm}^3$ , comparable to the value obtained on macroscopic properties ( $300 \text{ nm}^3$ ).\*



# Boron speciation in the melt

High-Temperature NMR (CNRS CEMTHI, ORLEANS)



IRMN  
THC

Incident Beam creates a local heating (thermal spike) leading to a "localized" ultra-fast Melt/Quench event : **the supervitrification**

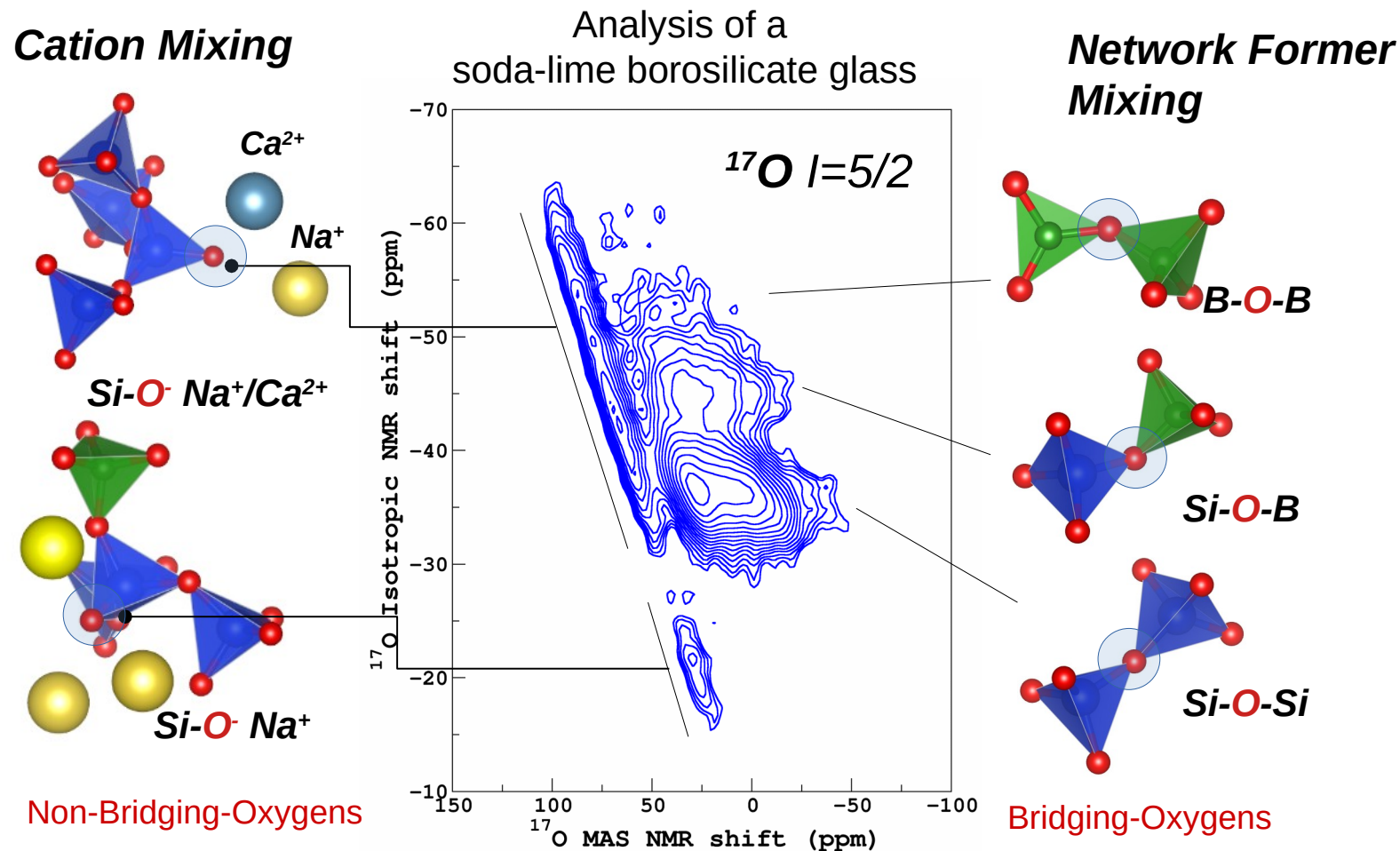
Massiot et al., Solid State Ion. **1990** 37 223-229  
 Bonafous et al., J. Chim. Phys. **1995** 92 1867-1870  
 Lacassagne et al., J. Phys. Chem. B **2002** 106 1862

C. Mendoza et al., JNCS **2014**, S. Peuguet et al. JNCS **2013**

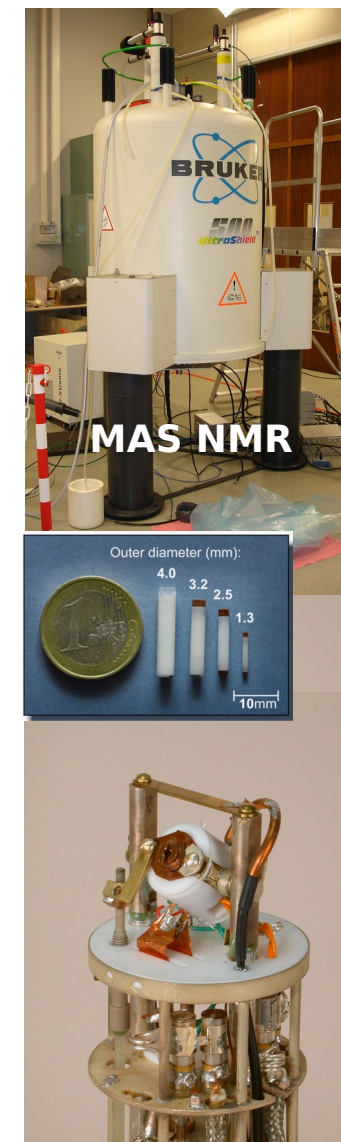


# MAS NMR in Oxide Glasses

Deciphering the Intermediate Range Order (IRO) :  $^{17}\text{O}$  2D MQMAS NMR



$^{17}\text{O}$  MQMAS NMR reveals the structural motifs at a **molecular** level:  
Accessing the **intermediate** range order (IRO)  
NMR offers other (many) possibilities based on through-space (J) and -bond (D) interactions (but they are complex experiment for quadrupolar nuclei ...)  
 $^{17}\text{O}$  enrichment must be  $\geq 40\%$

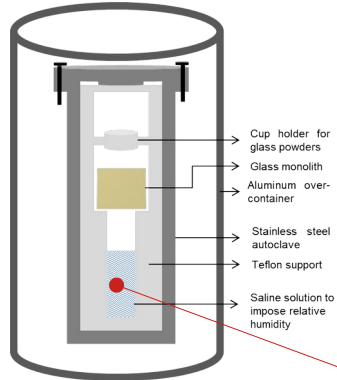


# NMR of altered Glasses: scrutating the altered layer

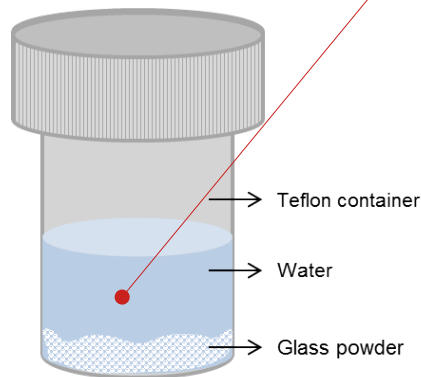
## The oxygen-17 NMR approach

S. Narayanasami et al., npj Mat. Deg. 6 (2022)

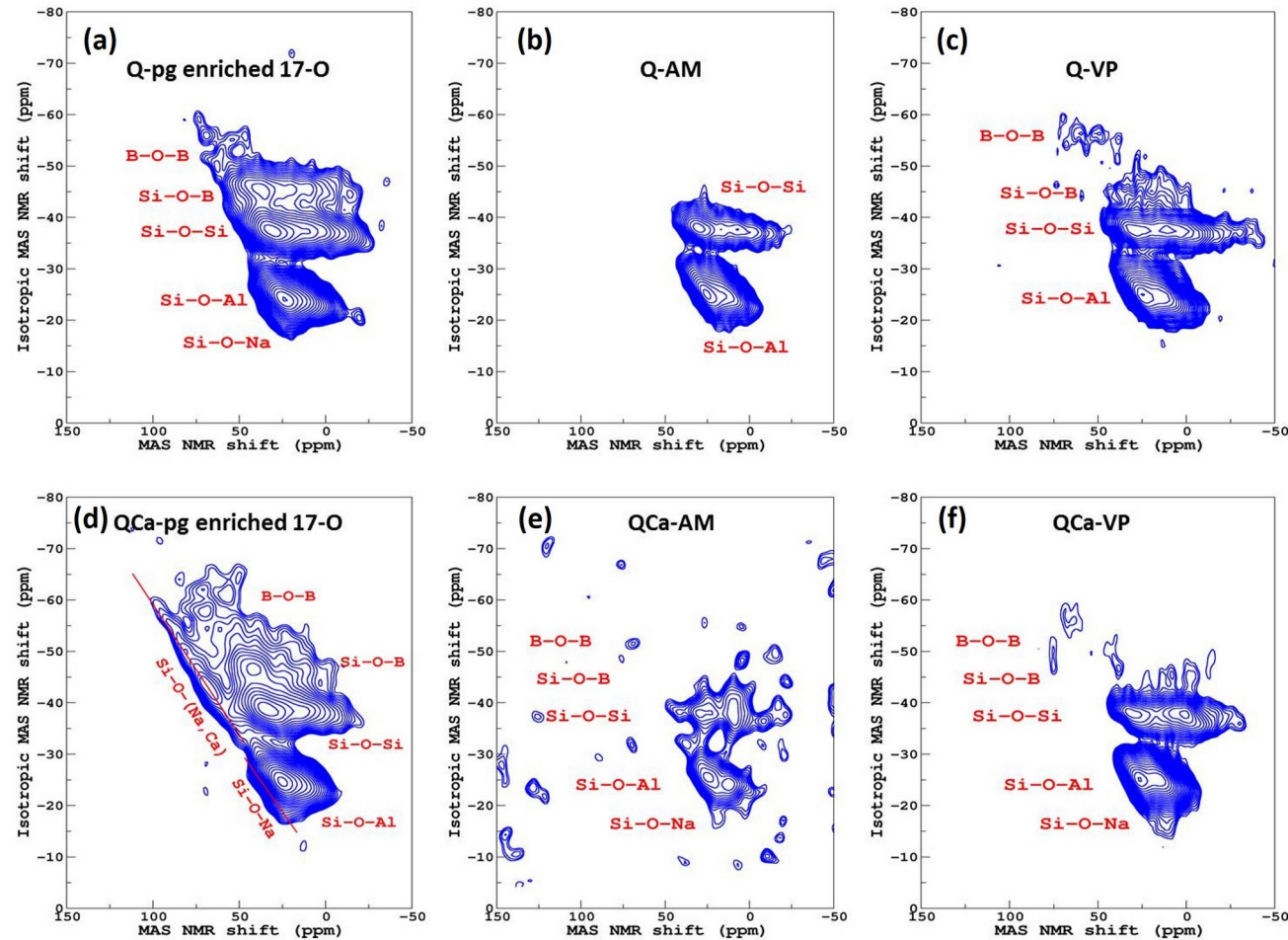
### Vapor Phase (VP)



### Aqueous Media (AM)



$^{17}\text{O}$   
>40%



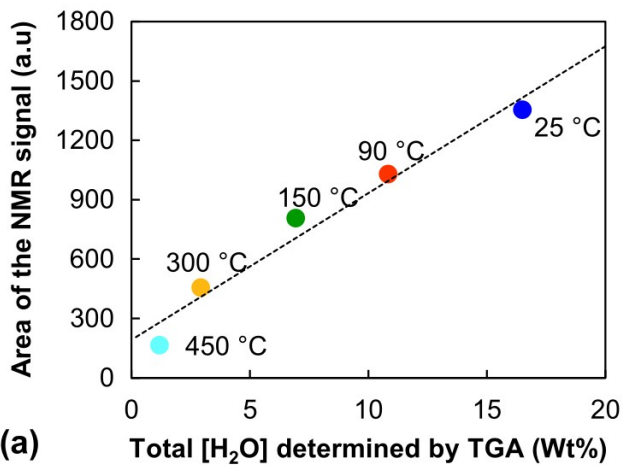
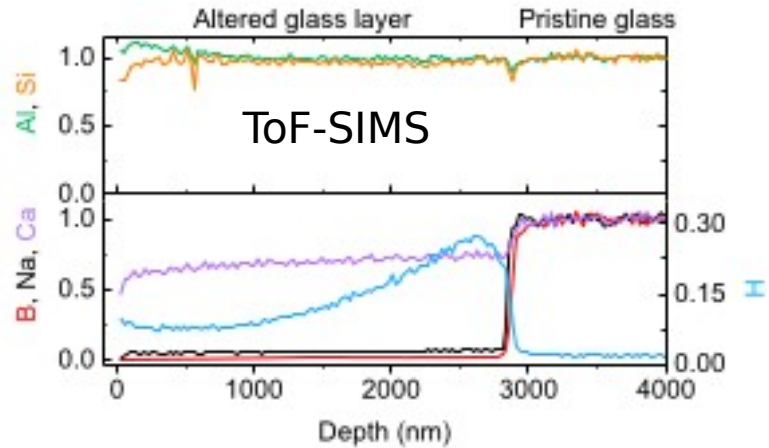
- Dominant Si-O-Si and Si-O-Al peaks (as expected...) in altered layer (depleted in boron and sodium)
- Confirmation of B-O-B and Si-O-B formation in VP altered samples only
- Weaker signal in QCa
- Note:  $^{17}\text{O}$  MQMAS filters out the (Si)-O-H signals



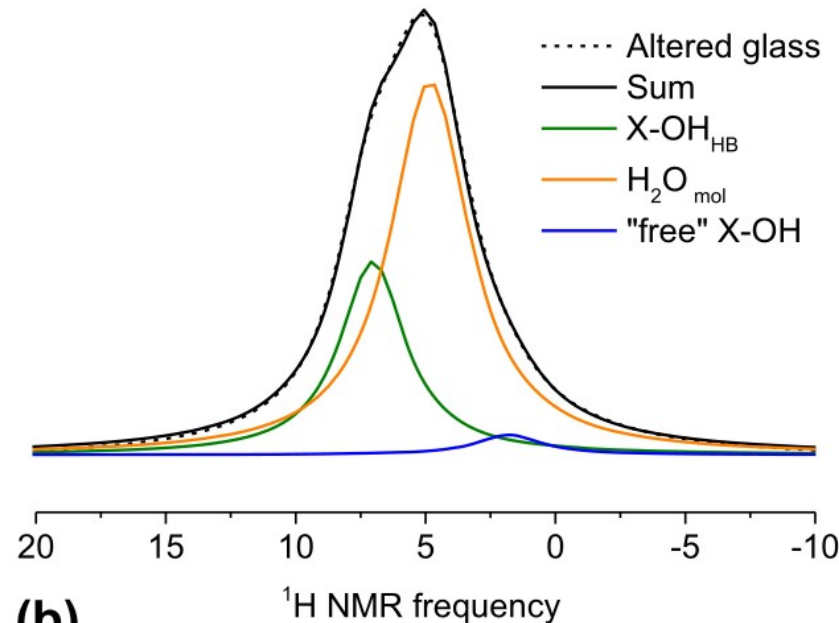
# NMR of altered Glasses: scrutating the altered layer

$^1\text{H}$  MAS NMR – accurate quantification of molecular water content

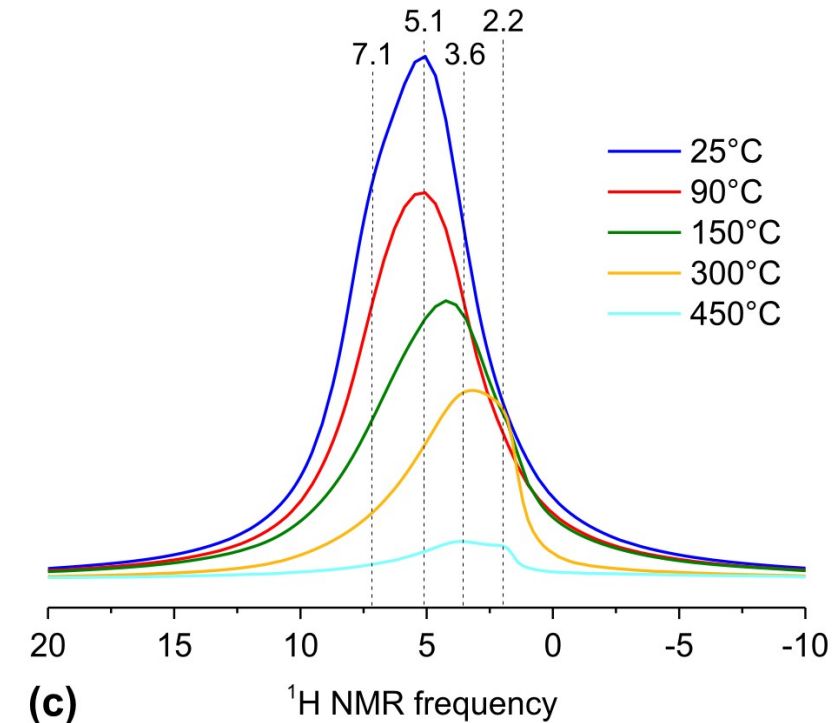
M. Collin et al., npj Mat. Deg. 4 (2018)



(a)



(b)



(c)

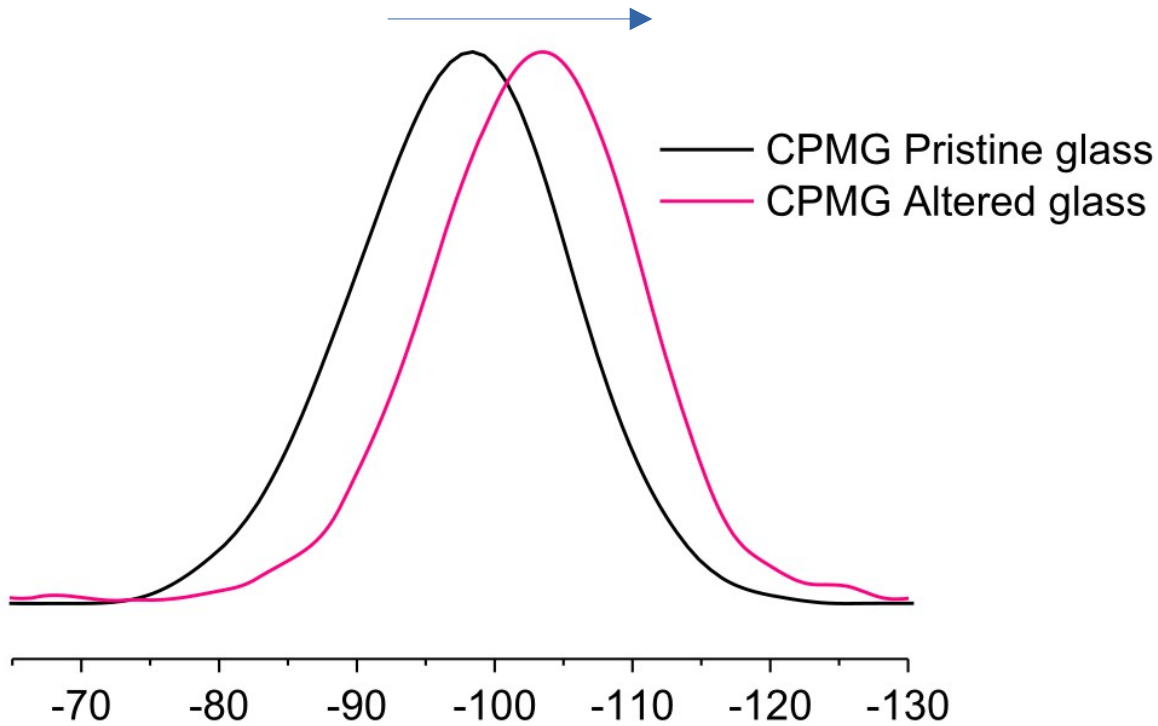
With molecular water removal, hydrogen bonds disappear, causing a decrease of the  $^1\text{H}$  shift of the Si-OH line (free Si-OH at  $\sim 2$  ppm)

Alteration at  $\text{pH}(90^\circ\text{C})=7$  in a solution saturated with amorphous silica (no silicate network dissolution). Glass alteration is therefore driven only by the transport of water. See M. Collin, npj Mat. Deg. 4 (2018)

# NMR of altered Glasses: scrutating the altered layer

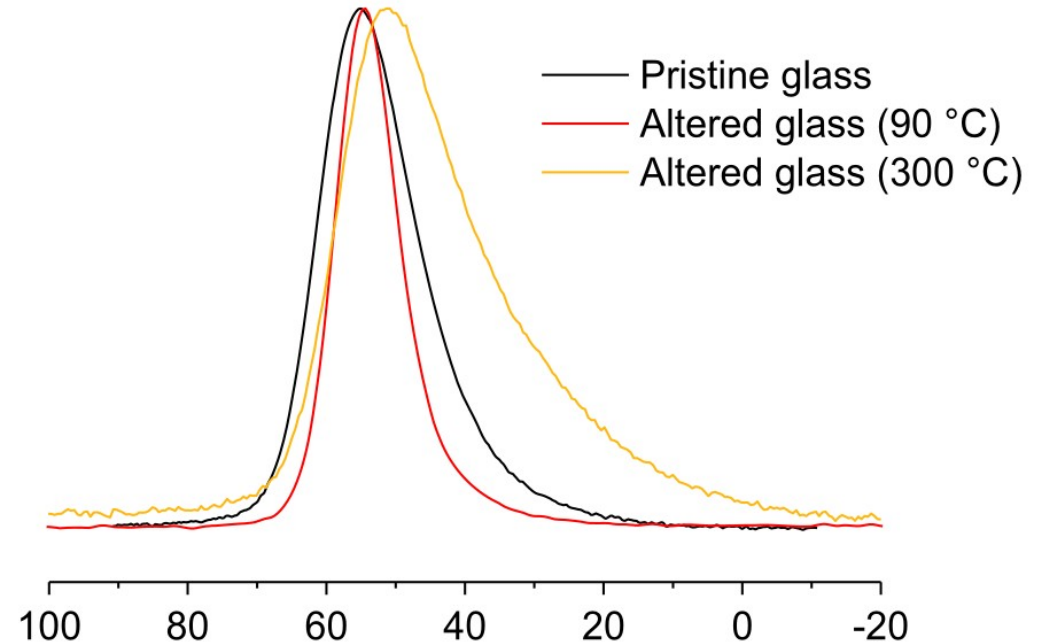
MAS NMR – modification of the network

M. Collin et al., npj Mat. Deg. 4 (2018)



(a) Solid NMR -  $^{29}\text{Si}$  chemical shift (ppm)

Variation due to a change in polymerization but also as a result of the **boron depletion**.

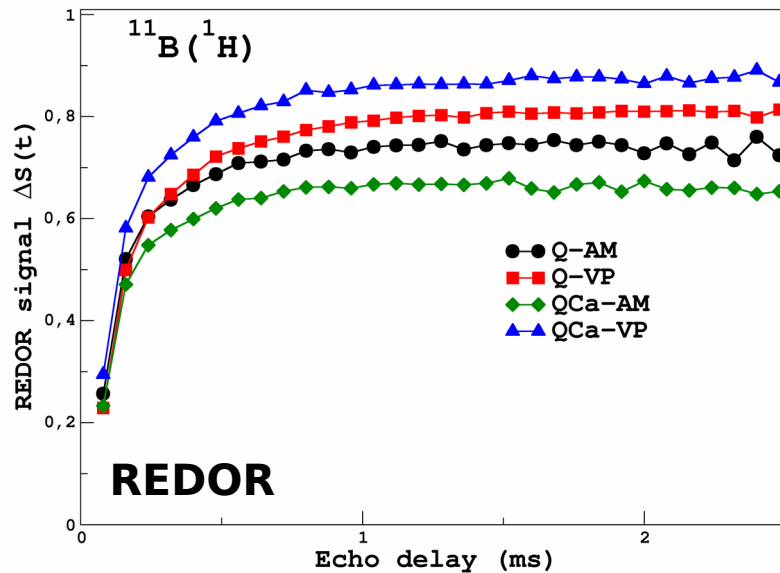
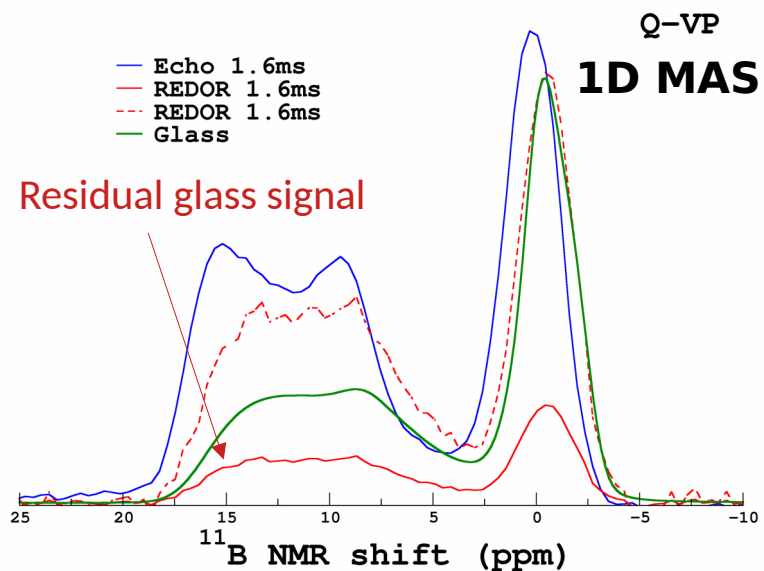


(c) Solid NMR -  $^{27}\text{Al}$  chemical shift (ppm)

In gel,  $\text{AlO}_4$  linewidth reduction due to coordination by  $\text{H}_2\text{O}$  water molecules. Removing molecular water, coordination by  $\text{Ca}^{2+}$ .  $\text{Ca}^{2+}$  is « solvated ».

# NMR of altered Glasses: scrutating the altered layer

$^{11}\text{B}$  MAS NMR – Probong the partial retention of B in the gel

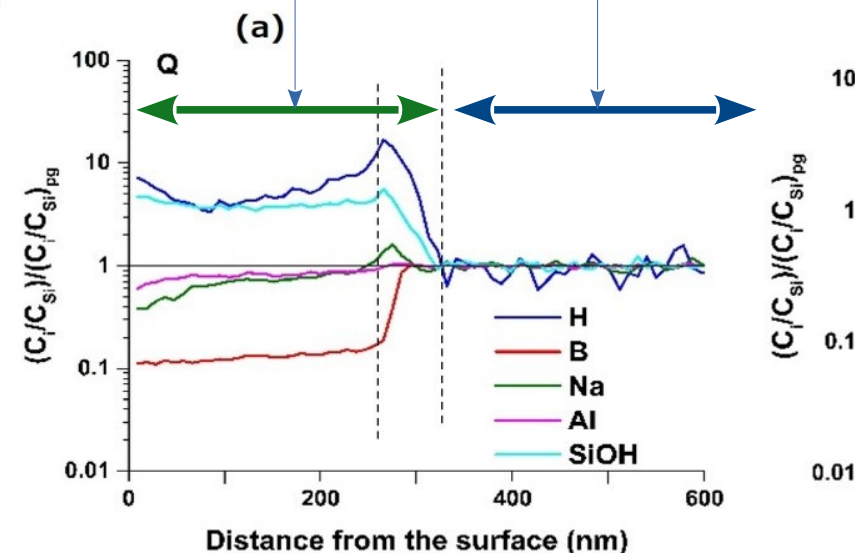
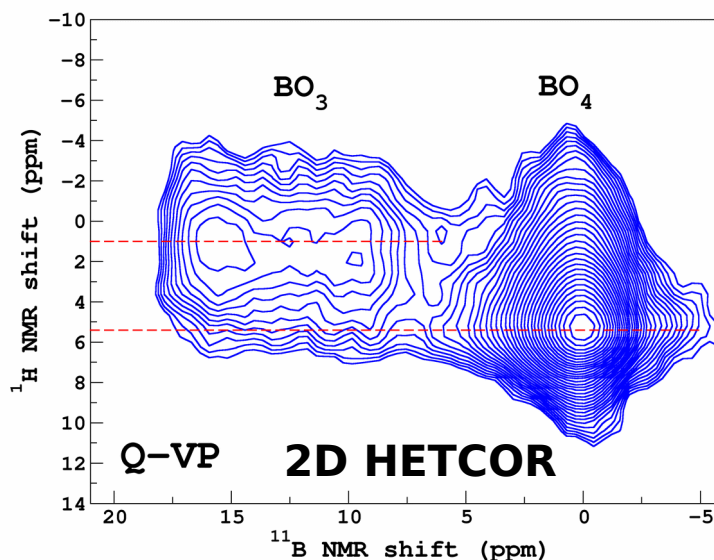


~ non-hydrated Boron (core glass)

~ hydrated Boron (altered glass)

NMR offers techniques that can select to the boron in the hydrated with  $^1\text{H}$ - $^{11}\text{B}$  Cross-Polarisation experiment (CPMAS) or REDOR, based on a transfer of magnetization between the two nuclei.

**S. Narayanasami** et al., npj Mat. Deg. 6 (2022)

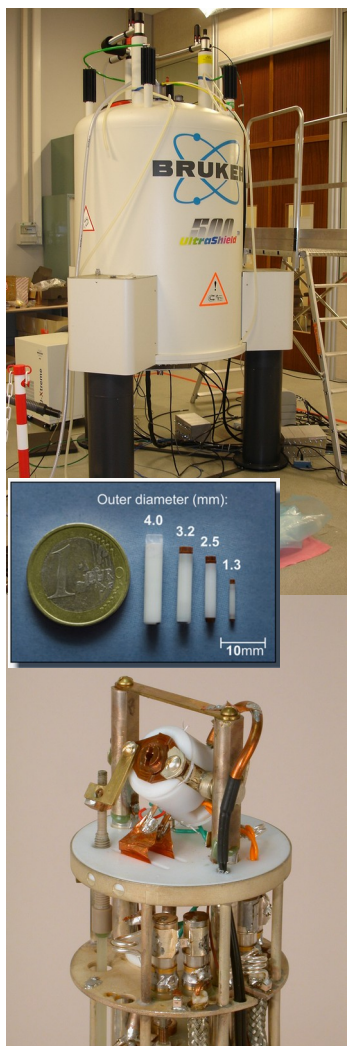


**(c)**

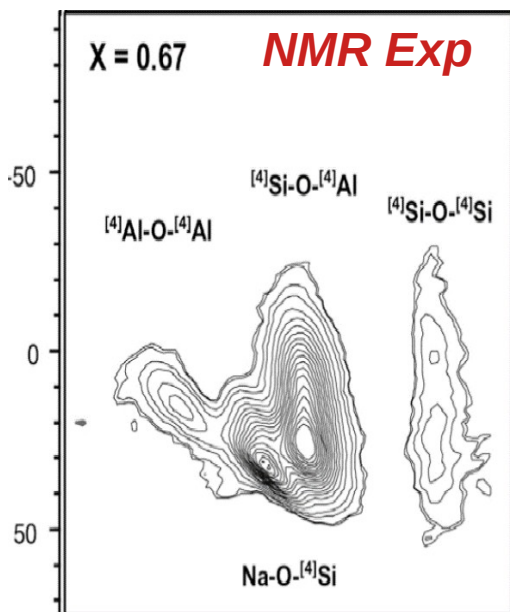


# The MD-GIPAW Approach

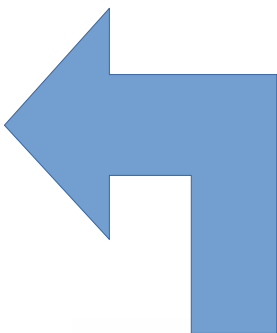
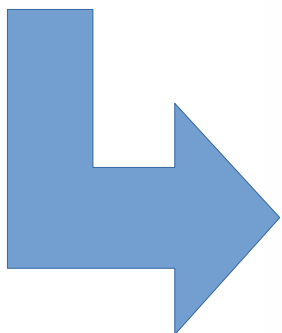
Computing NMR experiments from First-Principles



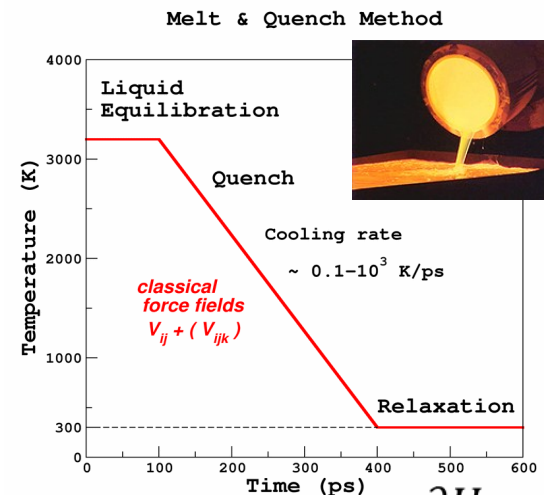
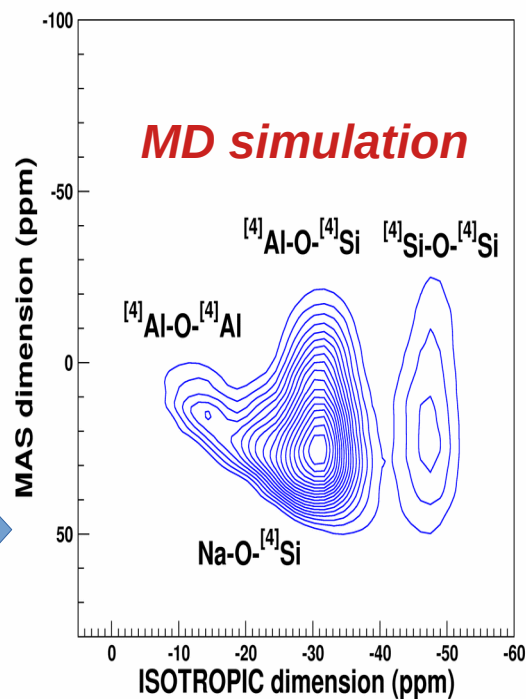
$^{17}\text{O}$  MQMAS, 11.7 T ~1 to 5 days



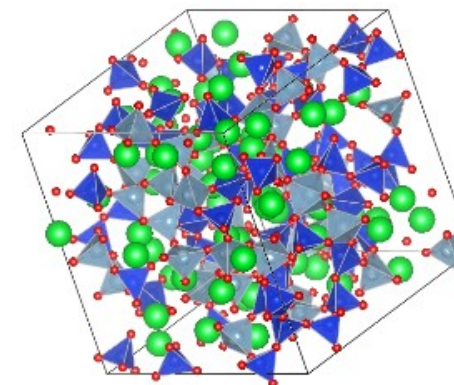
**Experiment Interpretation**



**MD assessment**



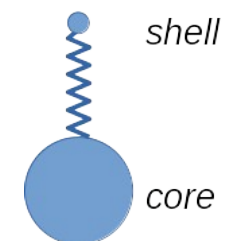
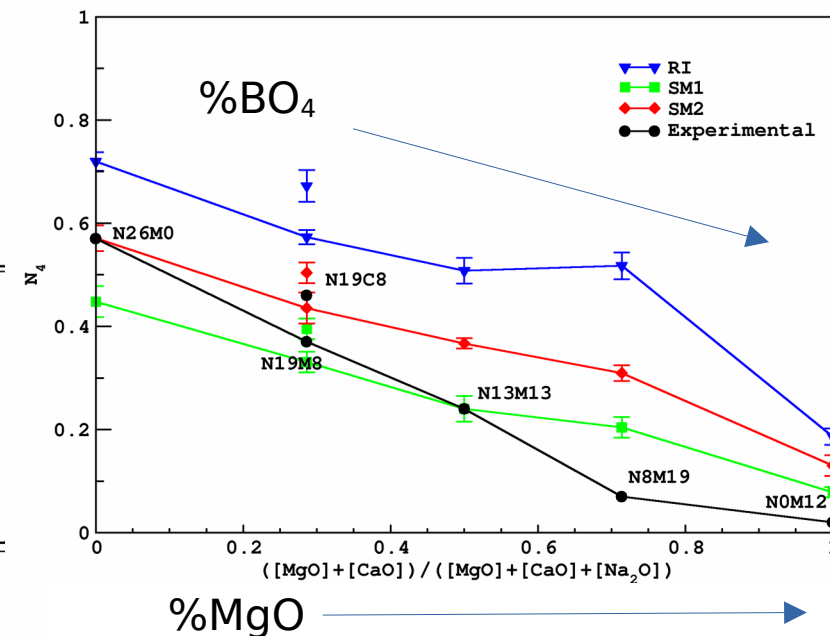
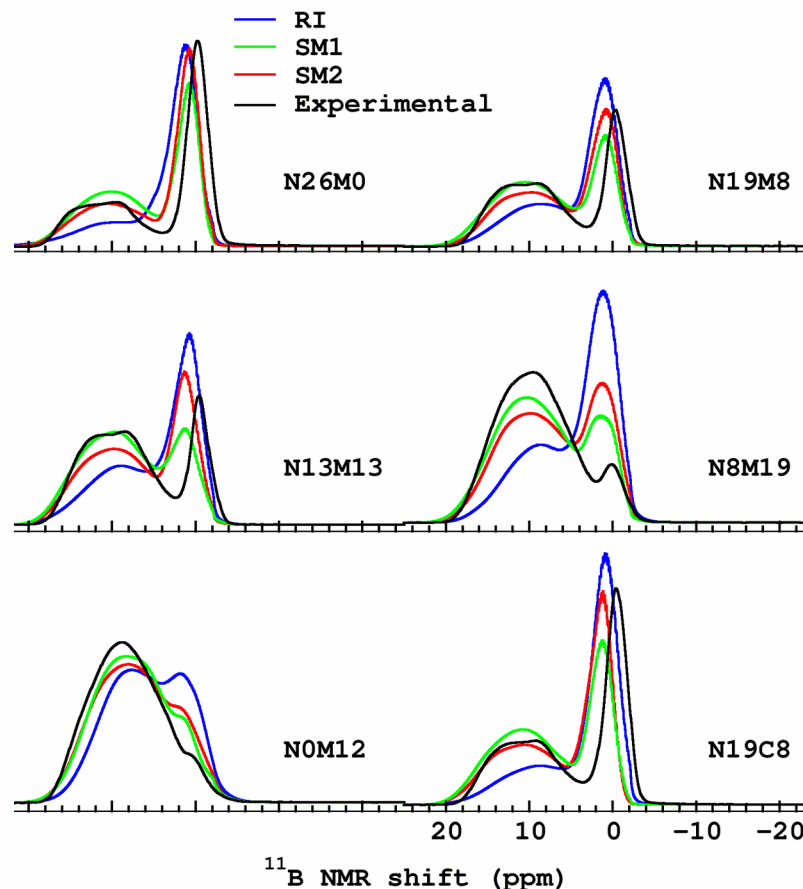
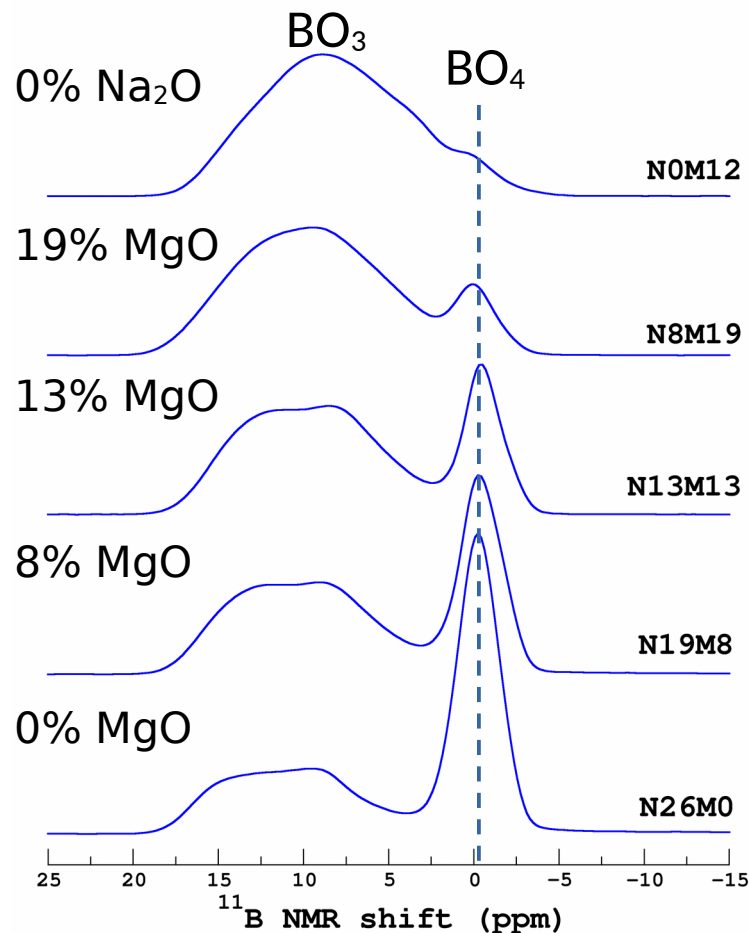
$$m_i \ddot{r}_i = F_i = - \frac{\partial U}{\partial r_i}$$



VASP-GIPAW, 400-800 atoms  
~ 4-8h, 48-96 procs

# Impact of Magnesium on NW glasses

Effects of Magnesium on the Structure of Aluminoborosilicate Glasses:  
**NMR Assessment of Interatomic Potentials Models for Molecular Dynamics**

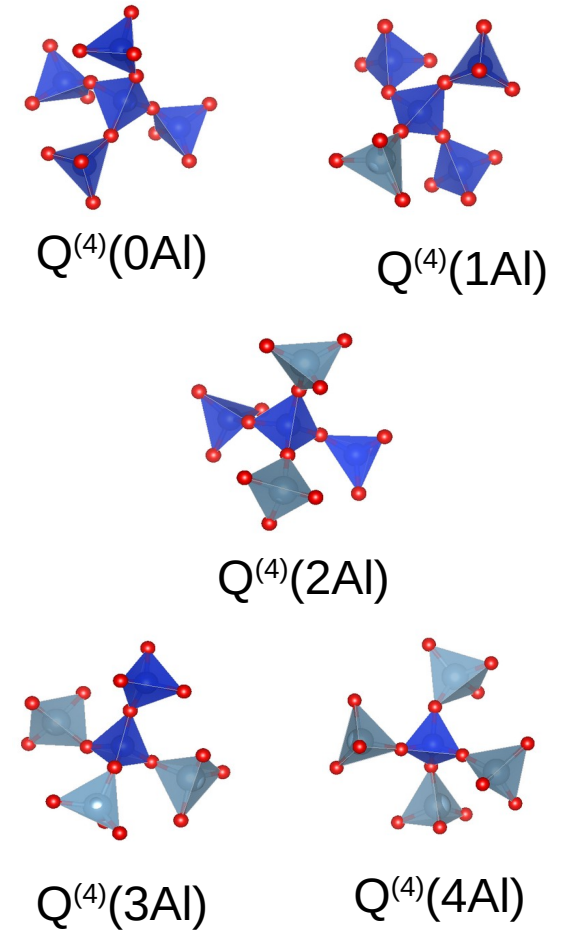
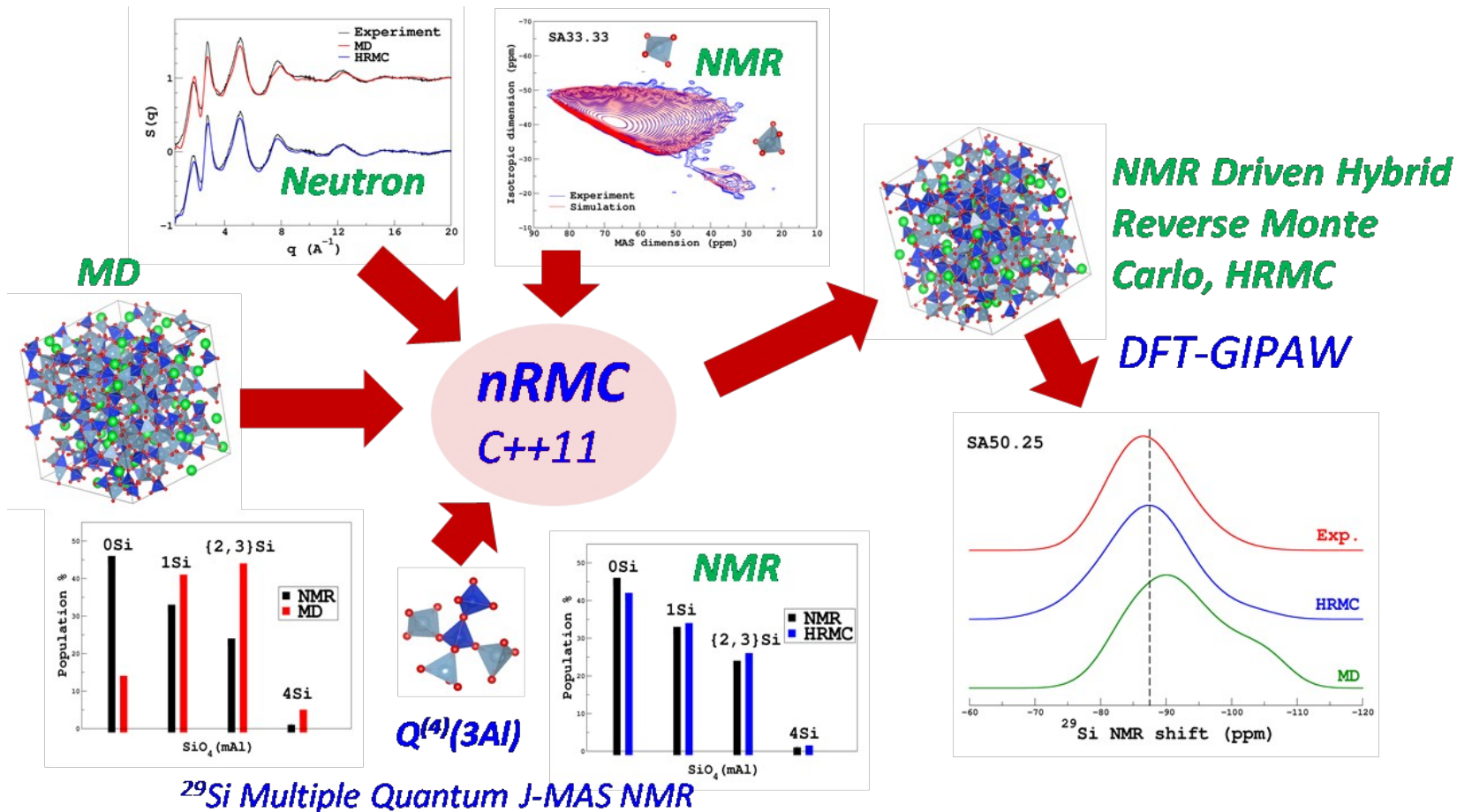


*Drude Polarizable  
Oxygen Model*

*Shell Model (polarizable oxygen atoms) performs much better than Rigid Ion Model (RI) to reproduce NMR features*

# NMR-Driven Reverse Monte Carlo

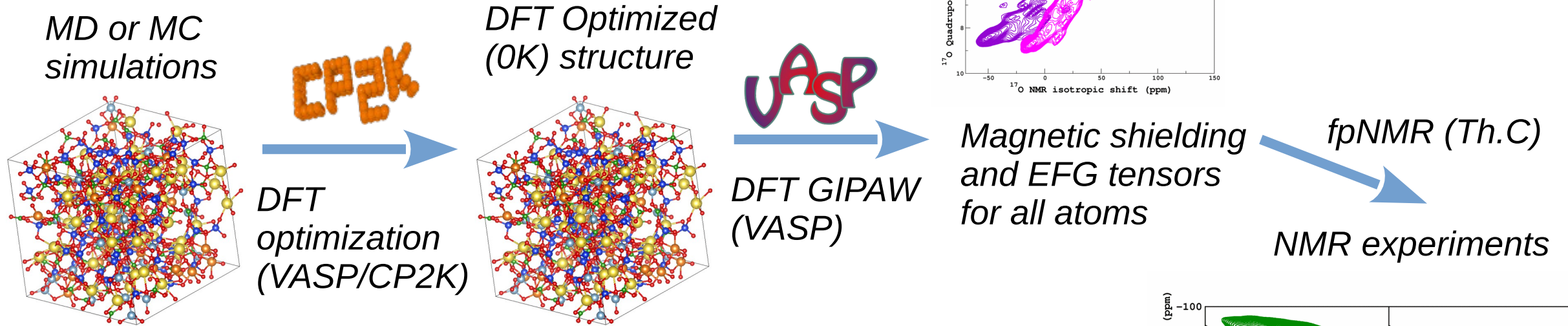
Including NMR constraints (& shifts?) in RMC Modelling



Using a **RMC** approach including **IRO constraints** (here  $Q^{(n)}(mAl)$  units) in the modelling significantly **improves the NMR simulations** (here  $^{29}\text{Si}$  MAS NMR spectra) and macroscopic properties (such as density)



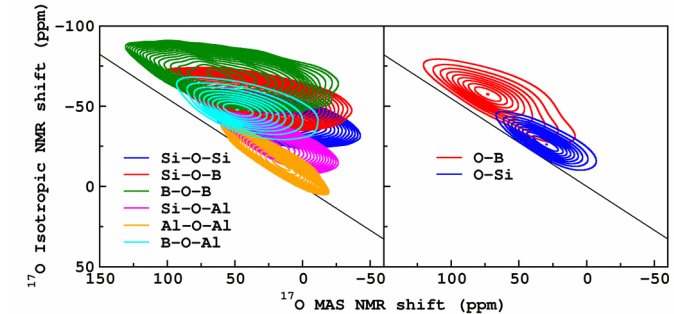
# The MD-GIPAW workflow



Current limitations in size due to the high cost of DFT computations  
CP2K / VASP ~ 1000 atoms.

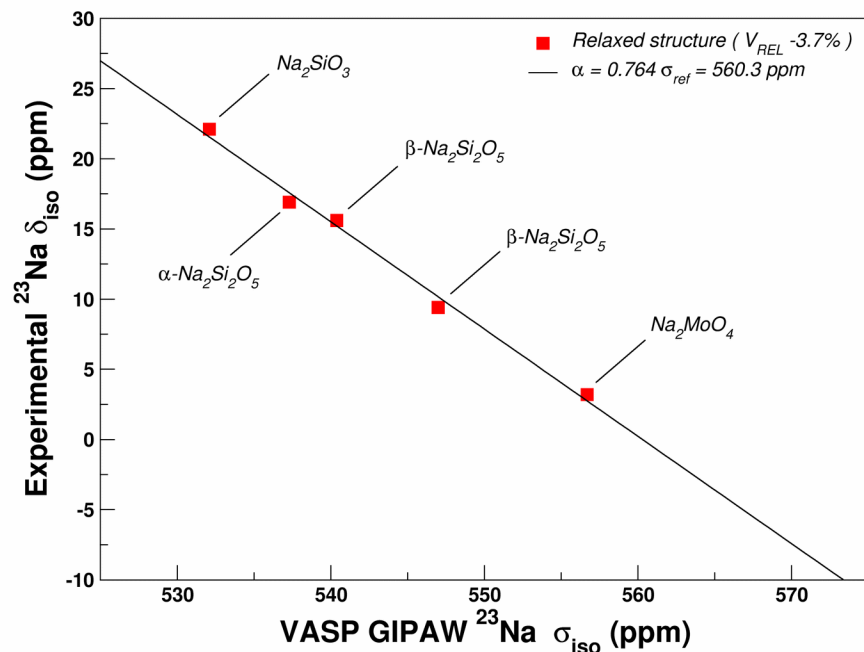
MD-GIPAW Modelling is limited to **frozen structures** : vibrations and local mobilities are not accounted for that can impact the experimental NMR spectrum (motional narrowing effects).

Solution : use **Machine Learning** !



# The DFT GIPAW method

Computing NMR properties in periodic solids : C.J. Pickard, F. Mauri, Phys. Rev.



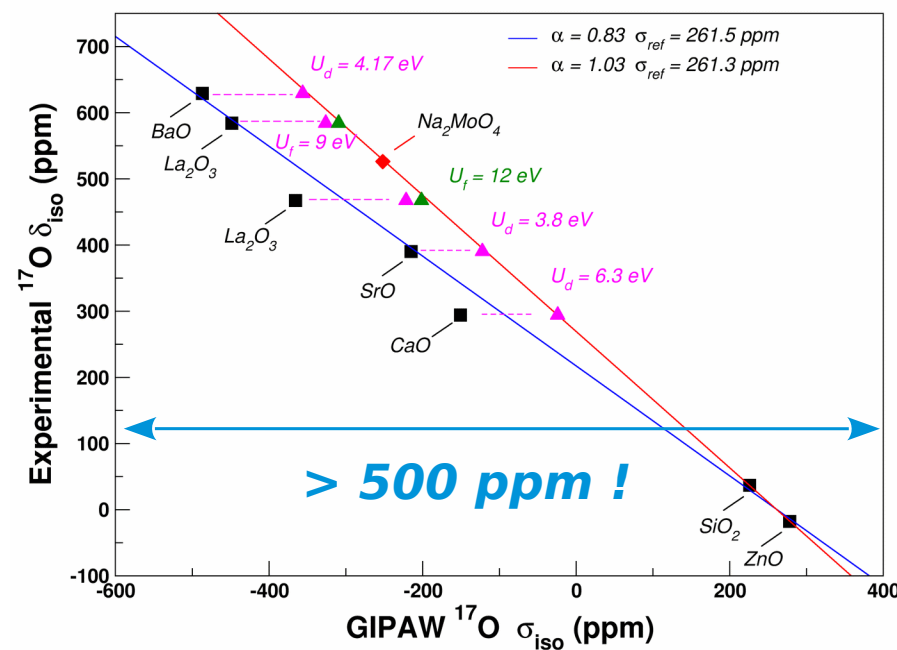
- Plane Wave Pseudopotential Method
- Outstanding accuracy ( $\sim$  experimental accuracy)
- CASTEP, Quantum Espresso, VASP, (CP2K)
- Up to 1000 atoms systems (VASP)
- Typically  $\sim$  5h/48 procs for 400 atoms
- DFT+U required for (d,f) empty orbitals

## GIPAW REVIEWS :

T. Charpentier, *The PAW/GIPAW approach for computing NMR parameters: A new dimension added to NMR study of solids*, Solid State Nucl. Magn. Reson., **2011**, 40, 1-20.

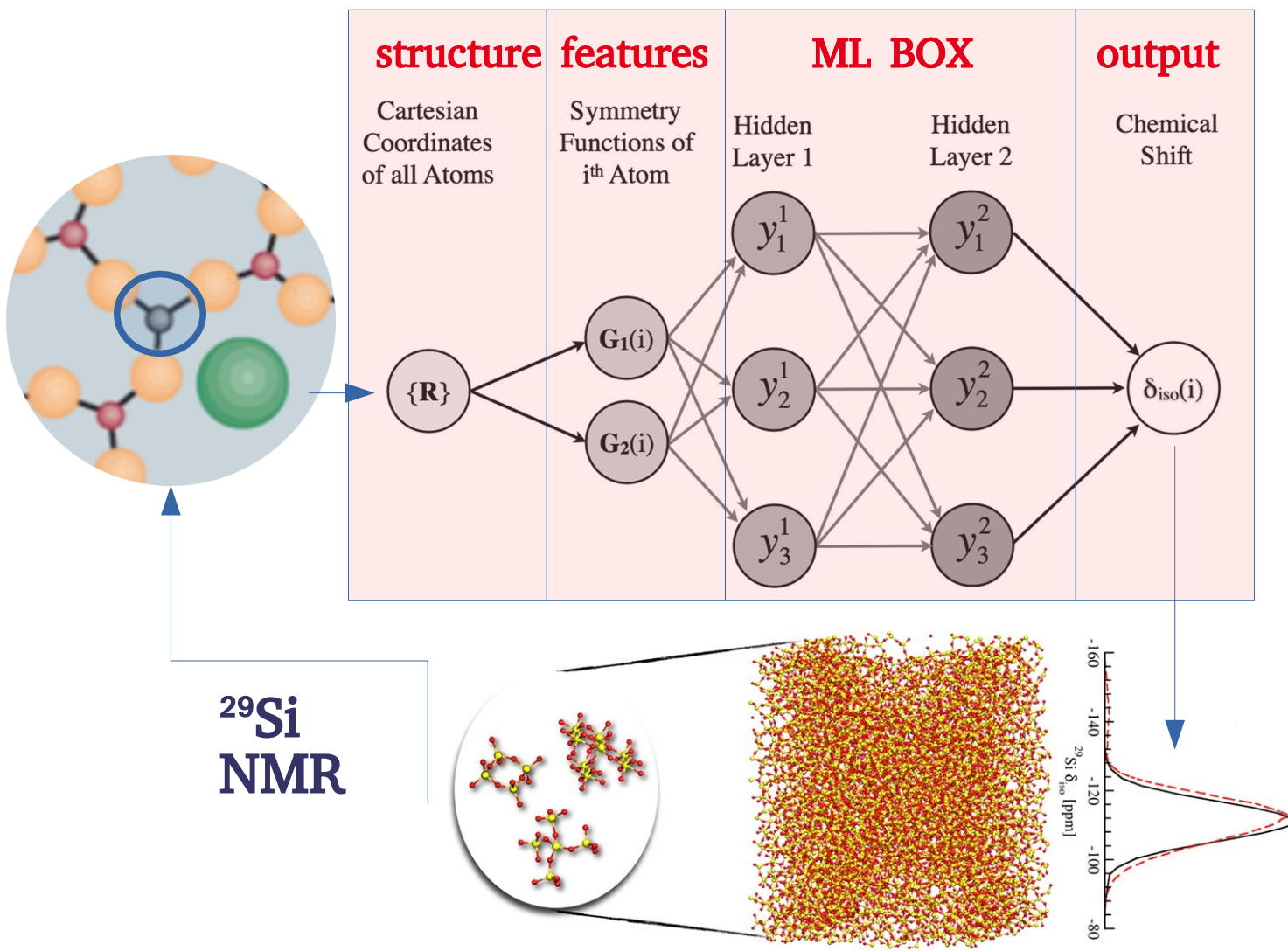
C. Bonhomme et al., *First-Principles Calculation of NMR Parameters Using the Gauge Including Projector Augmented Wave Method: A Chemist's Point of View*, Chem. Rev., **2012**, 112, 5733-5779.

T. Charpentier, A. Pedone, A.C. Menziani, *Computational simulations of solid state NMR spectra: a new era in structure determination of oxide glasses* ; RSC Adv. **2013**, 3, 10550-10578.

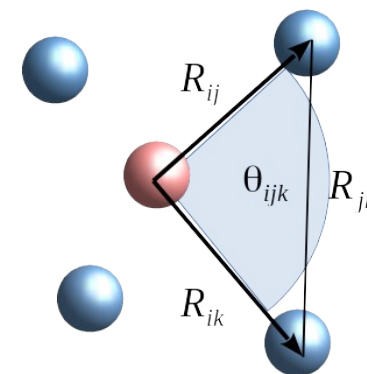


# NMR shifts by Machine Learning

A case of study : Vitreous Silica SiO<sub>2</sub>



## Atomic Descriptors (or features)



### Radial descriptors

$$G_i^{(1)}(\eta, R_s) = \sum_{j \neq i} e^{-\eta(R_{ij} - R_s)^2} f_c(R_{ij})$$

### Angular descriptors

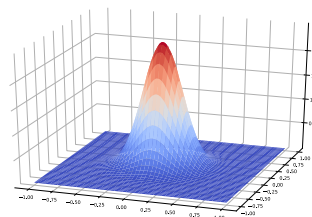
$$G_i^{(2)}(\eta, \lambda, \xi) = 2^{1-\xi} \sum_{j, k \neq i} (1 + \lambda \cos \theta_{ijk}) \times e^{-\eta(R_{ij}^2 + R_{ik}^2 + R_{jk}^2)} f_c(R_{ij}) f_c(R_{ik}) f_c(R_{jk})$$



# The SOAP descriptors

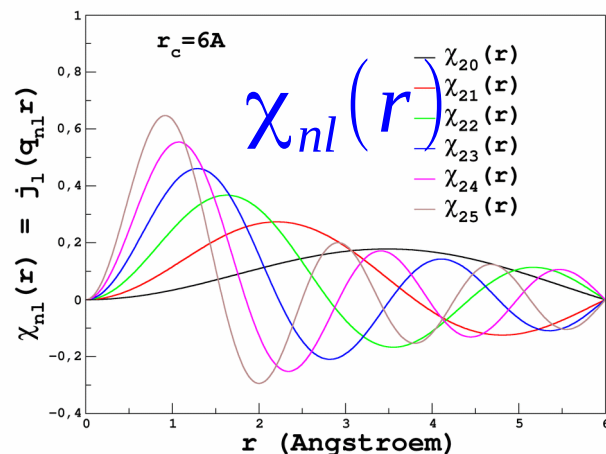
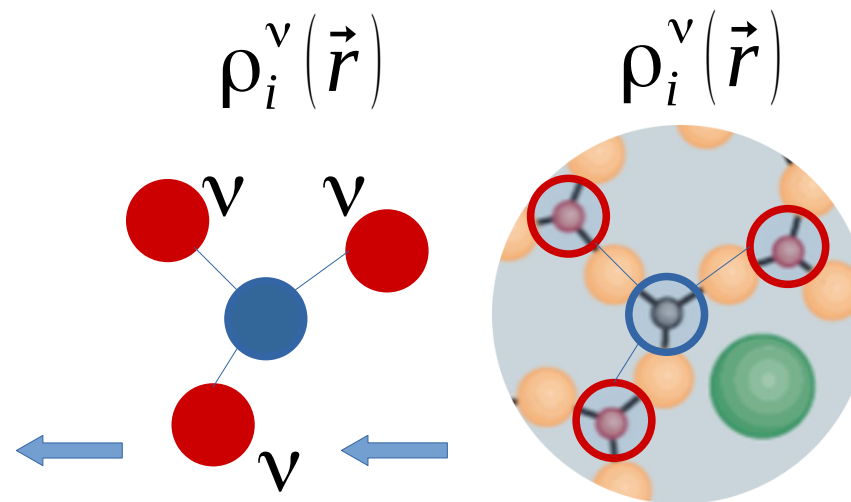
Smooth expansion of the atomic density with a radial cutoff function

$$\rho_i^v(\vec{r}) = \sum_{j \in N_i^v} G_\sigma(\vec{r} - \vec{r}_{ij}) f_c(r_{ij})$$



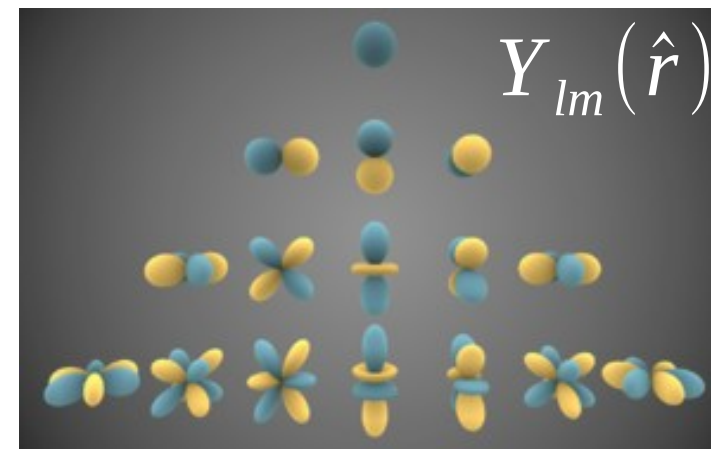
Spherical Harmonics & radial basis expansion

$$\rho_i^v(\vec{r}) = \sum_{nlm} c_{nlm}^v \chi_{nl}(r) Y_{lm}(\hat{r})$$



We found the spherical Bessel functions to be superior to other functions for the Radial Basis Expansion.

They form a natural orthonormal basis on the segment  $[0, r_c]$ .



A.P. Bartok et al. (2013) *On representing chemical environments*, Phys. Rev. B

E. Kocer et al. (2020) *Continuous and optimally complete description of chemical environments using Spherical Bessel descriptors*, AIP Advances 10, 015021

# The SOAP descriptors

Atomic local environment:

$$\rho_i^v(\vec{r}) = \sum_{nlm} c_{nlm}^v \chi_{nl}(r) Y_{lm}(\theta, \phi)$$

**Rotational invariant** (isotropic shift) descriptors are

$$c_{n00}^v$$

**2-body terms**

To get more descriptors, one option is to couple channels in such a way that they form **rotational invariants**:

$$c_{nl+m}^v \times c_{n'l-m}^u$$

This is the so-called **SOAP** powerspectrum

$$p_{nn'l}^{v\mu} = \sum_m c_{nl+m}^v \times c_{n'l-m}^u$$

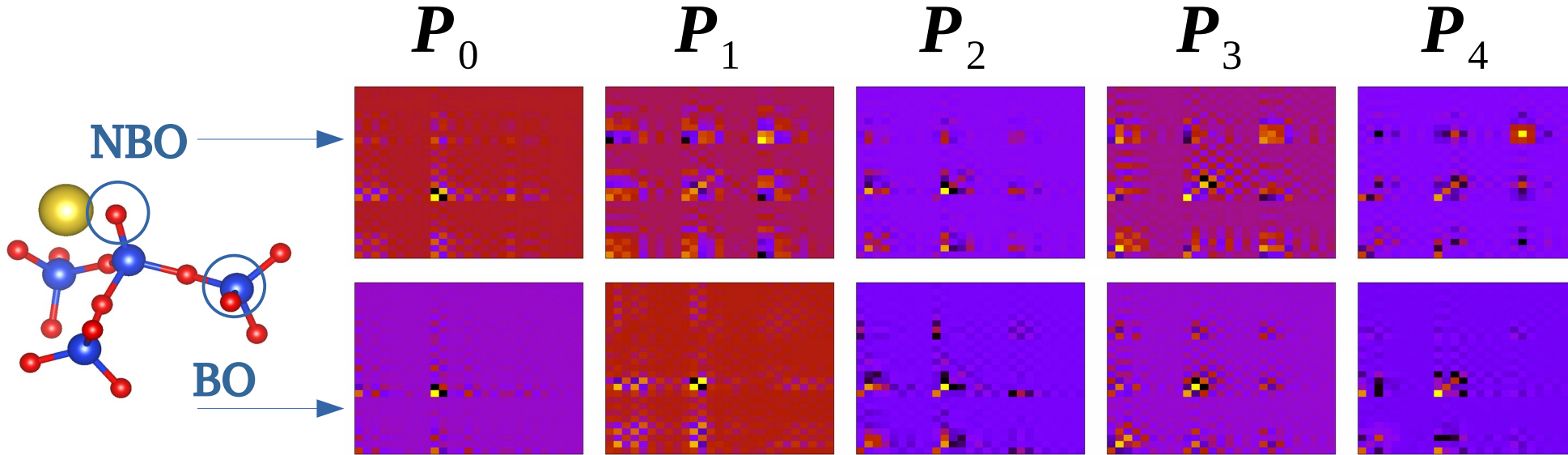
**3-body terms**

We can go farther... with the bispectrum

$$b_{nn_1n_2ll_1l_2}^{\lambda\nu\mu} = \sum c_{n_1l_1m_1}^\lambda c_{n_2l_2m_2}^\nu c_{n_1l_1m_1}^\mu c_{l_1m_1l_2m_2}^{lm}$$

**4-body terms**

# The SOAP descriptors



**n.b.** The powerspectrum can be seen efficiently stored using a Gram matrix product

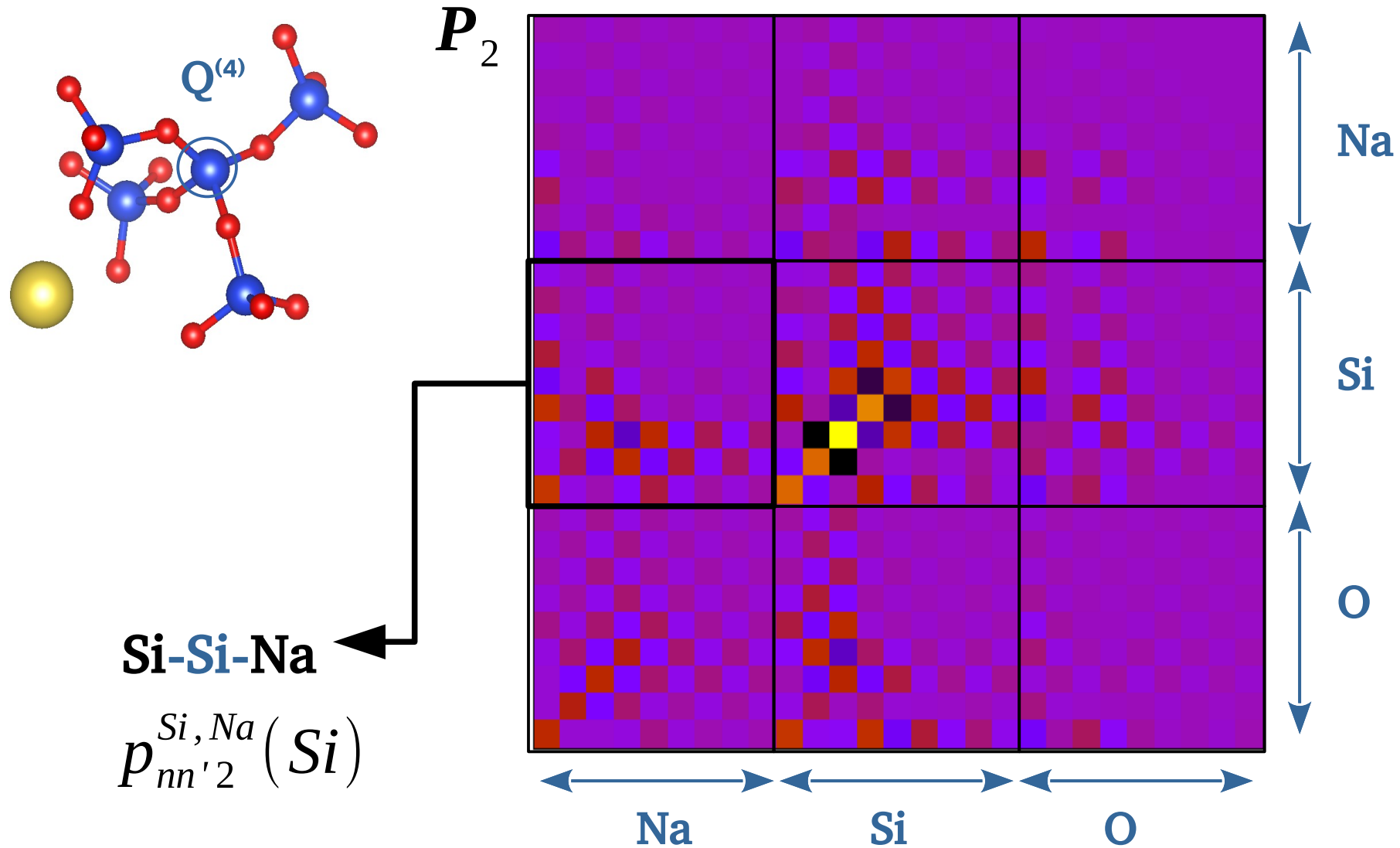
$$\mathbf{c}_{n,l}^{\mu} = \begin{pmatrix} \mathbf{c}_{n,l,-l}^{\mu} \\ \vdots \\ \mathbf{c}_{n,l,+l}^{\mu} \end{pmatrix}$$

$$\{p_{nn'l}^{\nu\mu}\} = \mathbf{P}_l = \mathbf{X}_l^t \cdot \mathbf{X}_l \text{ with } \mathbf{X}_l = \left( \mathbf{c}_{1,l}^{\mu} \mathbf{c}_{2,l}^{\mu} \dots \mathbf{c}_{N,l}^{\mu} \mathbf{c}_{1,l}^{\nu} \dots \mathbf{c}_{N,l}^{\nu} \right)$$



# The SOAP descriptors

In binary  $\text{Na}_2\text{O}-\text{SiO}_2$  glasses

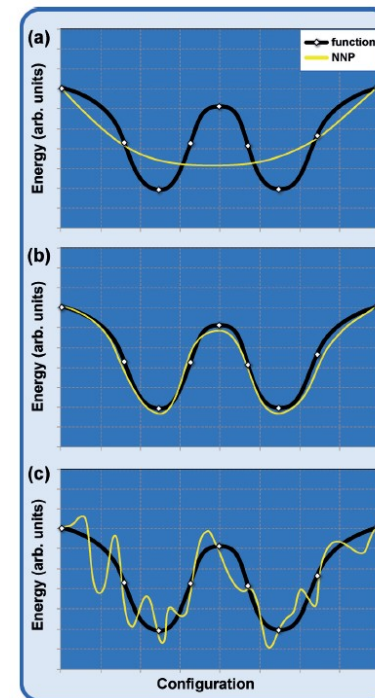
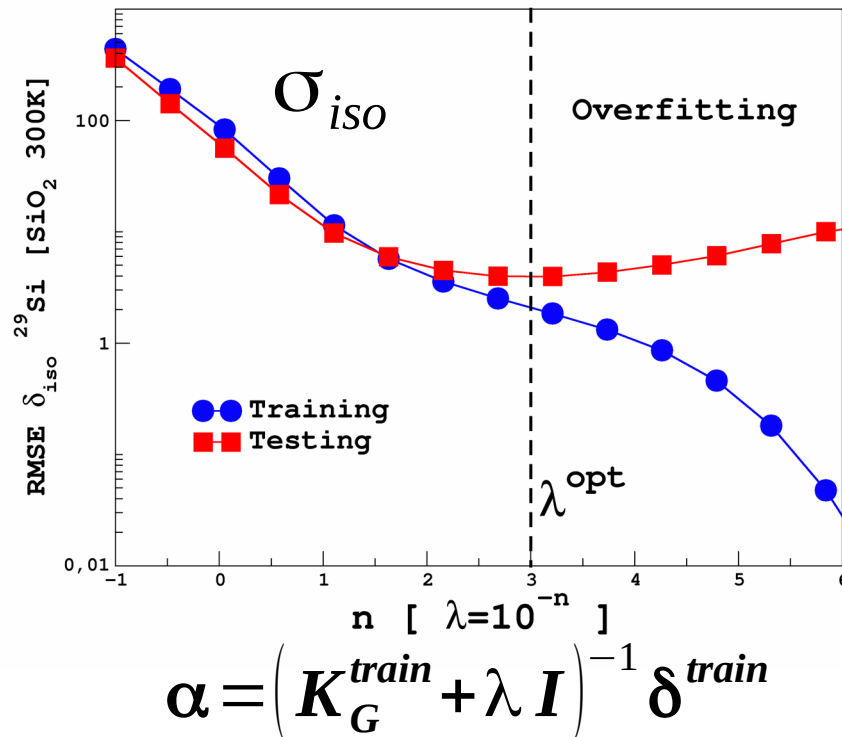
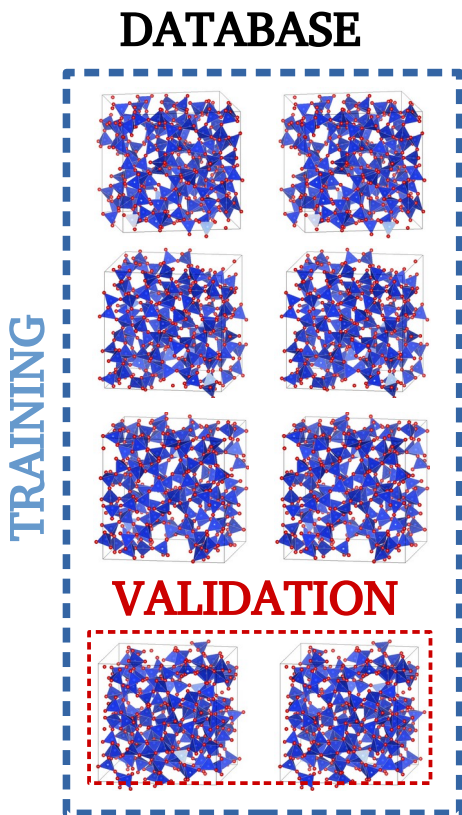


# NMR shifts by Machine Learning

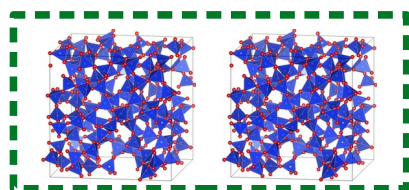
The Training Procedure : k-fold Cross-Validation



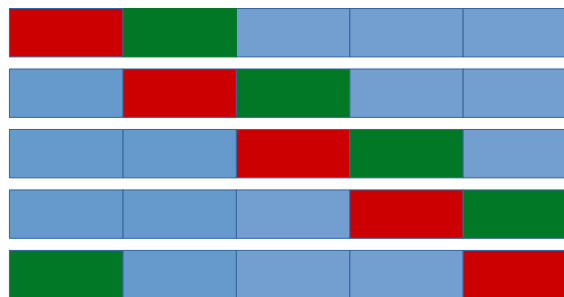
J. Behler et al. Angewandte Chemie 2017



$\lambda$  and  $\kappa$  are optimized on a validation set before testing.

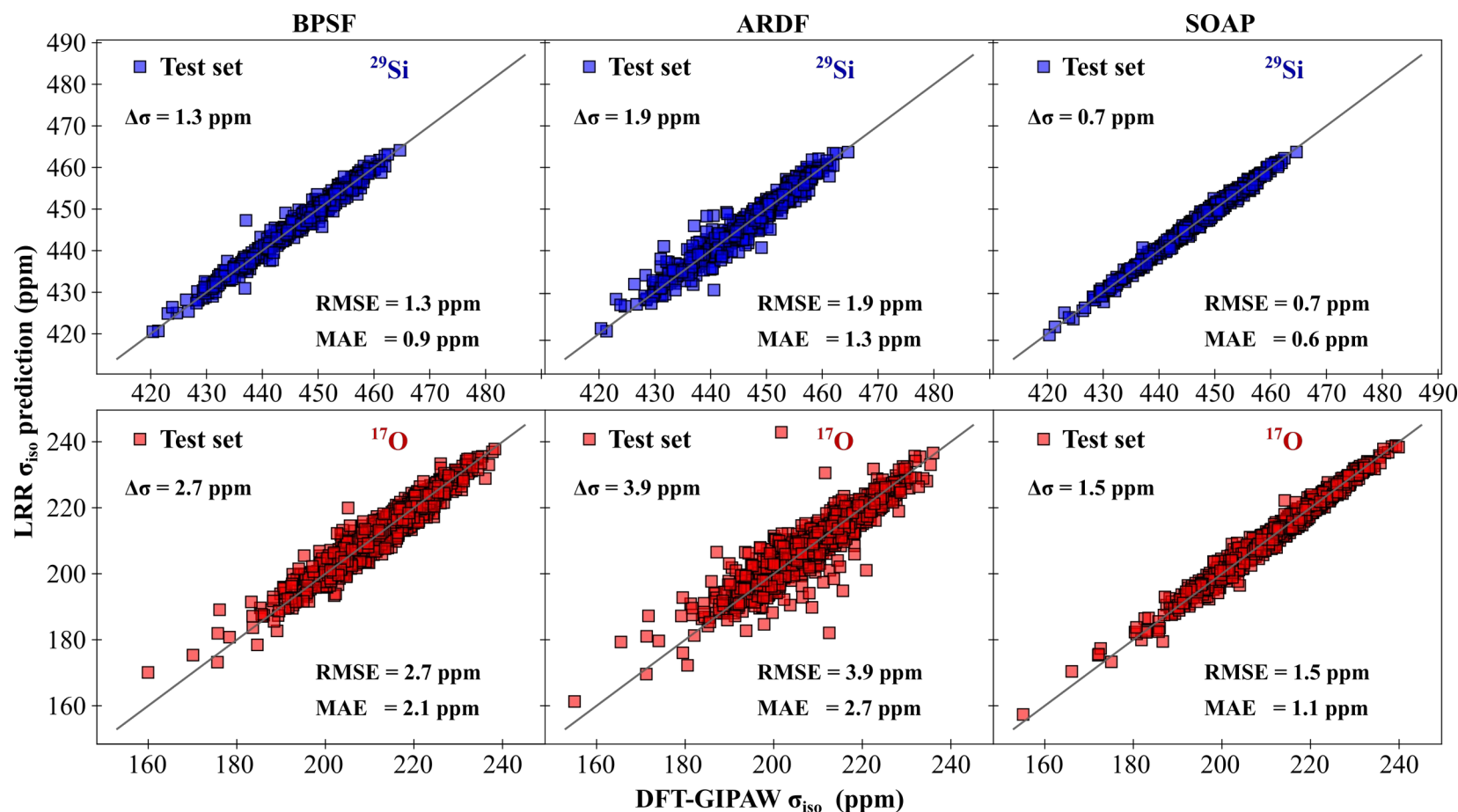


k-fold Cross Validation



# NMR shifts by Machine Learning

## Aluminosilicate Glasses : Comparison of Descriptors

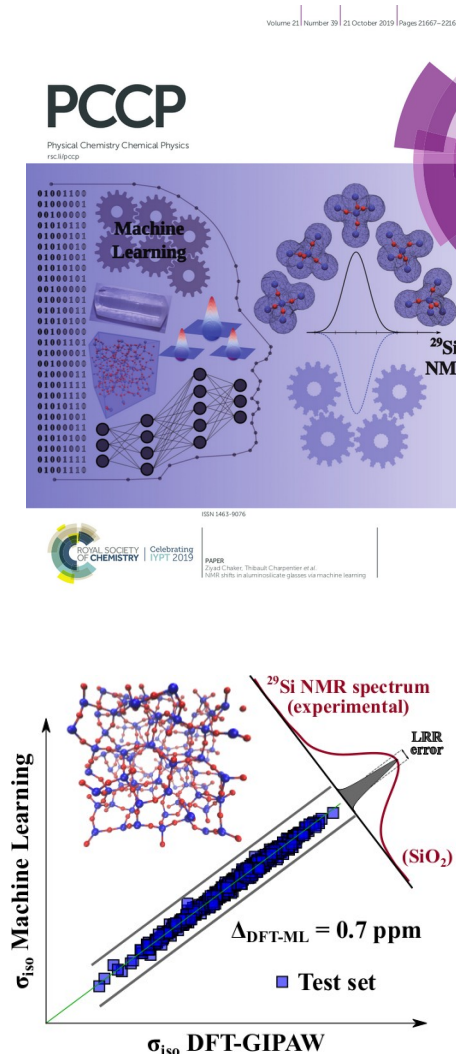


☞ SOAP (Smooth Overlap of Atomic Positions)

Symmetry Functions (BPSF) : J. Behler and M. Parrinello, *Phys. Rev. Lett.* 98, 146401 (2007)

Angular-Radial Distribution Functions (ARDF) : N. Artrith, A. Urban, and G. Ceder, *PHYSICAL REVIEW B* 96, 014112 (2017)

\*Smooth Overlap of Atomic Positions (SOAP) : Albert P. Bartók, Risi Kondor, and Gábor Csányi, *Phys. Rev. B* 87, 184115 (2013)

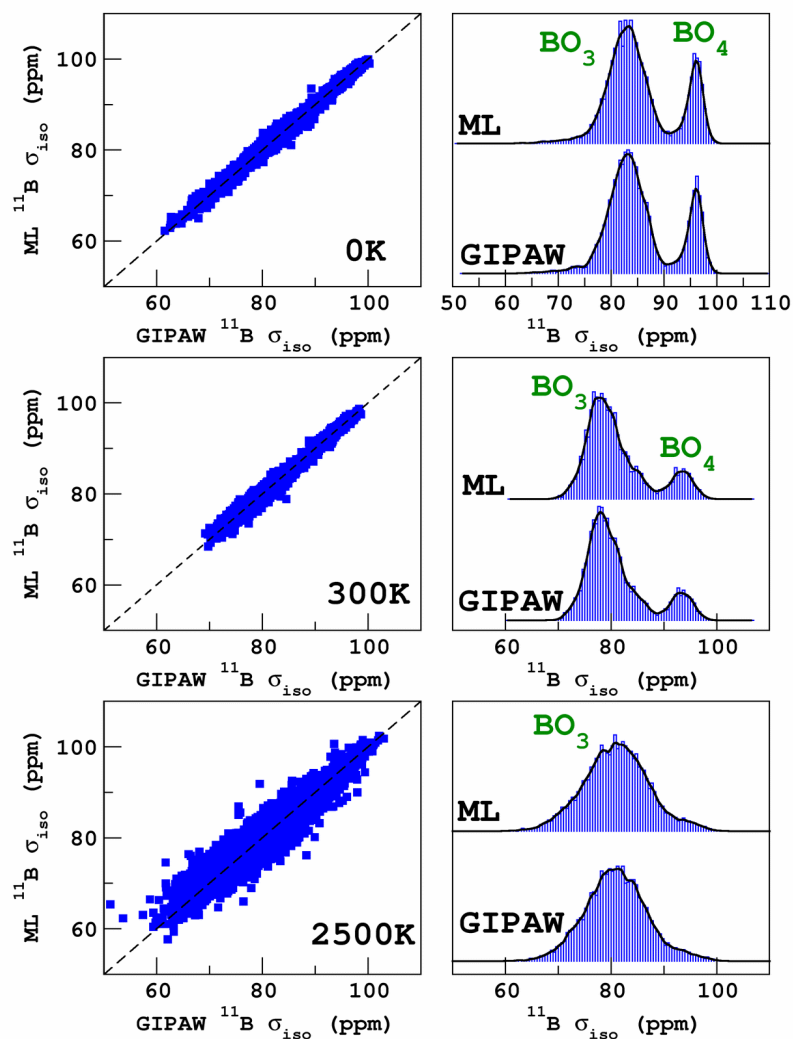


Z. Chaker et al. PCCP 2019

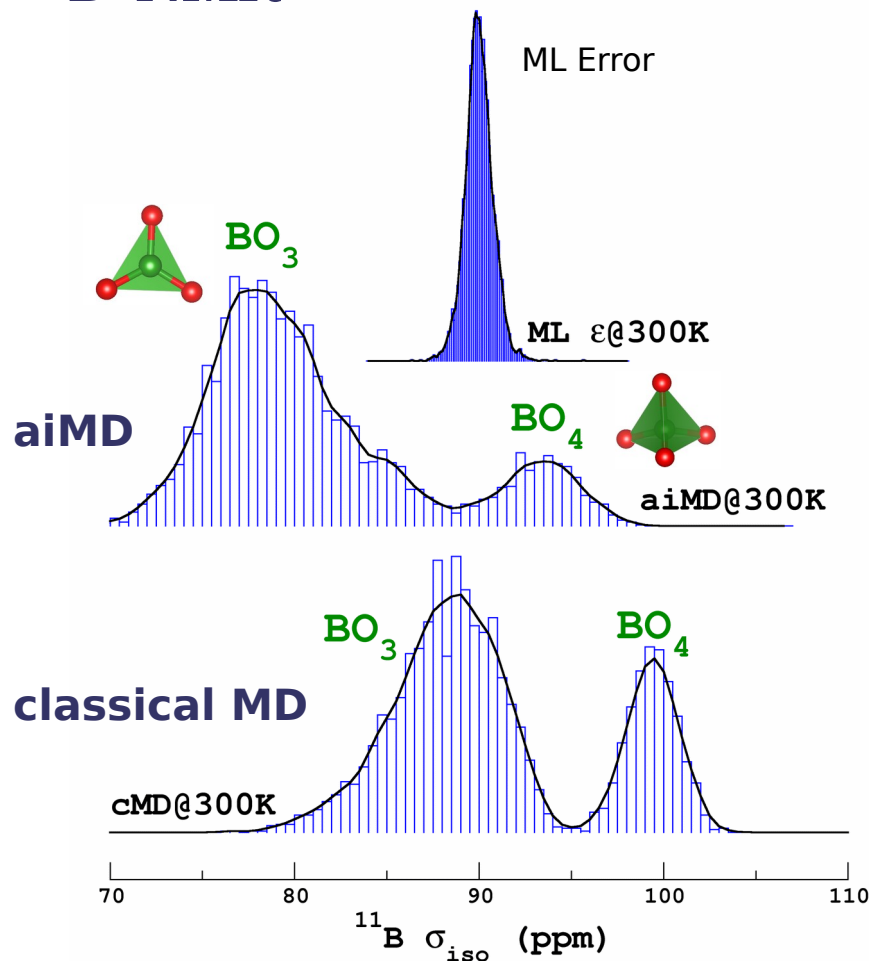


# Boosting DFT/NMR with Machine Learning

JSOAP descriptors and Least-Squares Support Vector Regression



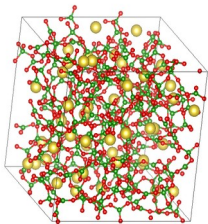
## $^{11}\text{B}$ NMR



Prediction of  $^{11}\text{B}$  magnetic shielding tensor (chemical shift) and Electric Field Gradient tensor (Quadrupolar Parameters)

# Boosting DFT/NMR with Machine Learning

jSOAP descriptors and Least-Squares Support Vector Regression

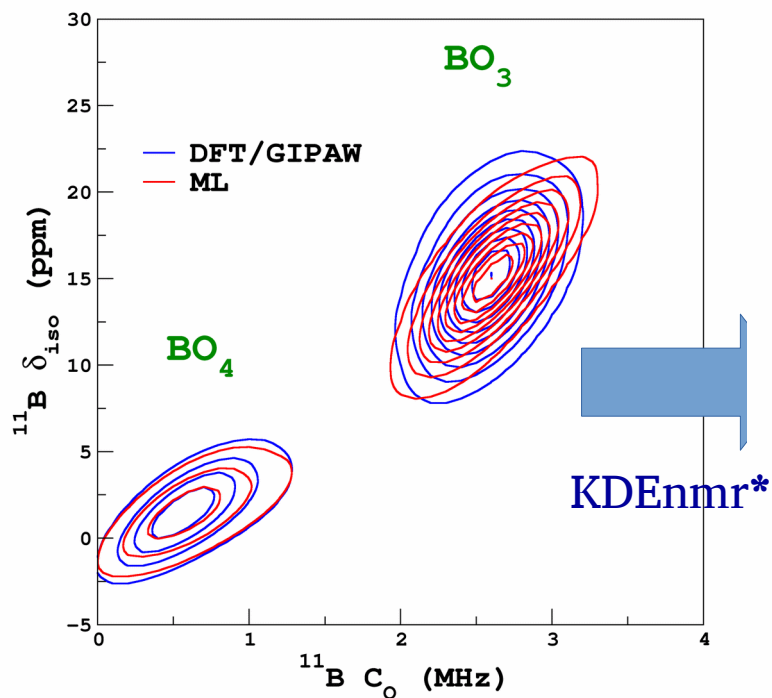


$^{11}\text{B}$  NMR

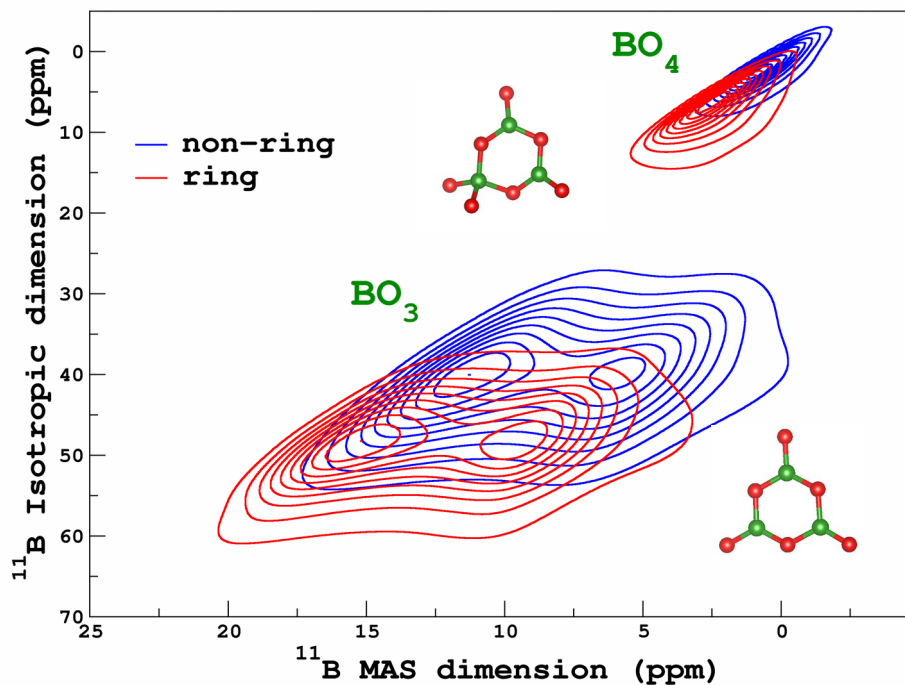
CPU for a single model of 800 atoms :

DFT ~ 96 (procs) \*8h

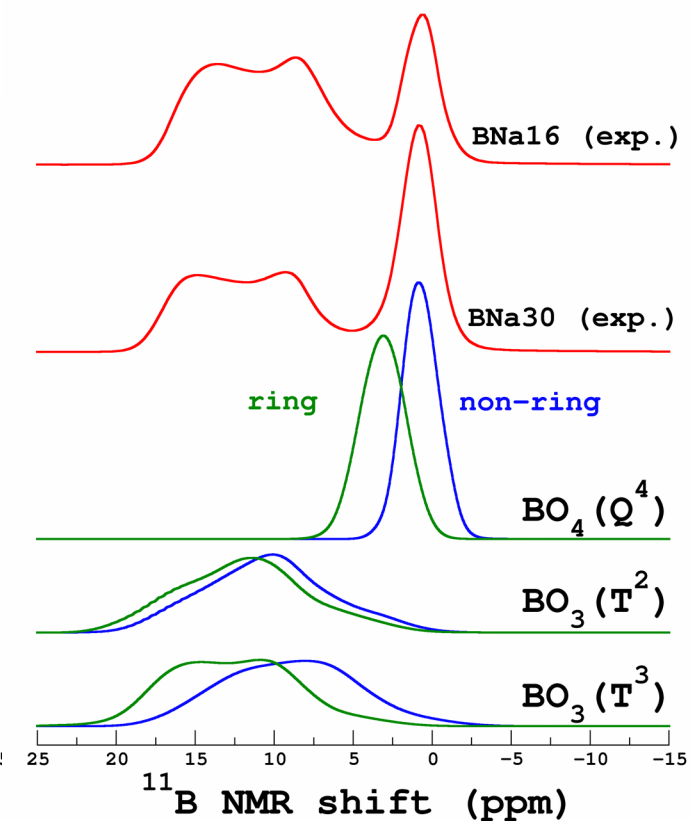
☞ ML ~ 10s (1 proc)



NMR parameter distribution



$^{11}\text{B}$  MQMAS NMR simulation



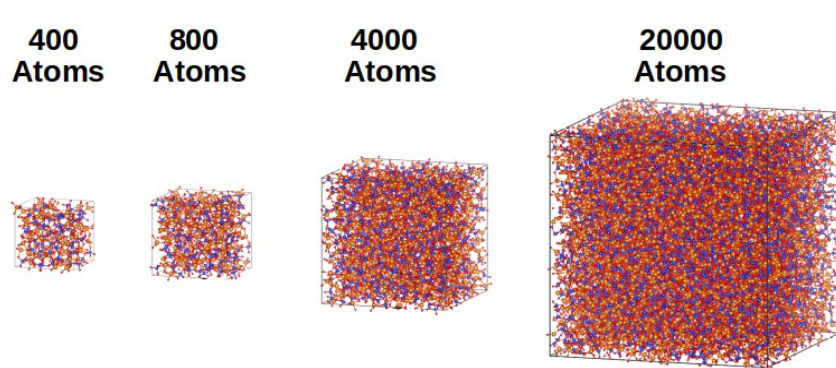
$^{11}\text{B}$  NMR fingerprints in Sodium Borate Glasses

\*Kernel Density Estimate of NMR parameter distribution

\*T. Charpentier et al. *First-Principles Nuclear Magnetic Resonance Structural Analysis of Vitreous Silica*, The Journal of Physical Chemistry C 2009 113 (18), 7917-7929.

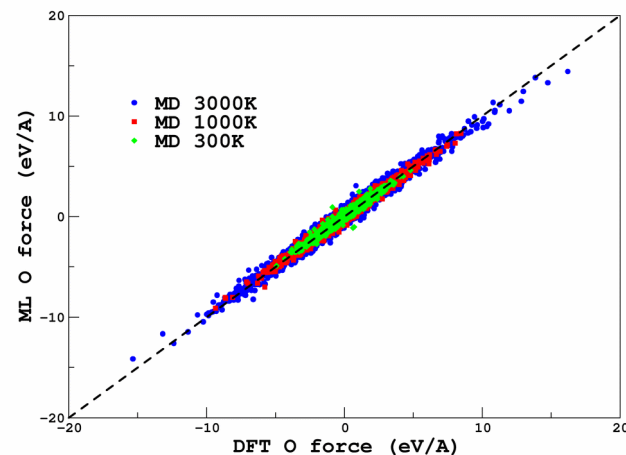
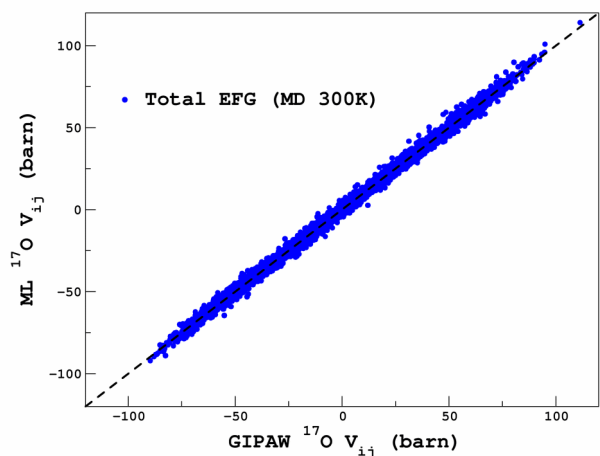
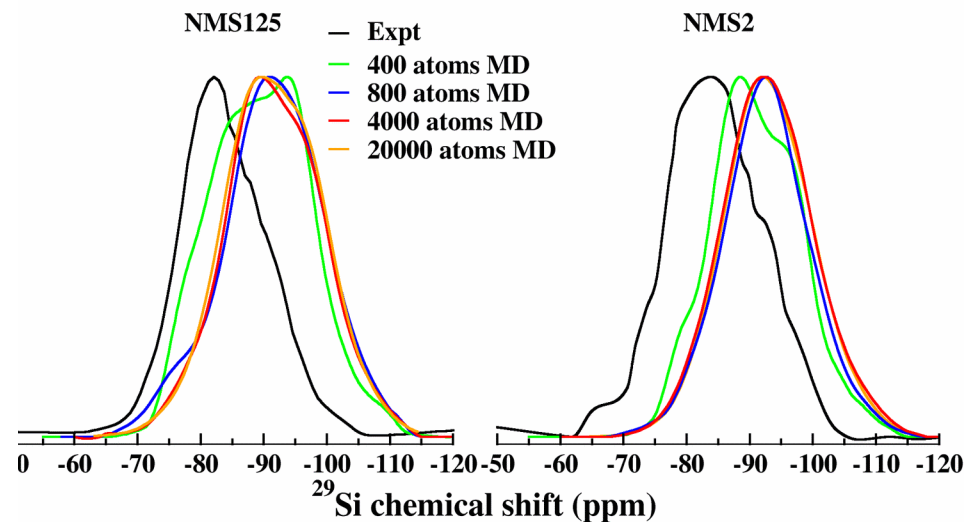
# Boosting DFT/NMR with Machine Learning

JSOAP descriptors and Least-Squares Support Vector Regression



DFT calculation time, 1 structure, 40 cores  
48h    1 week    ...    ∞ (too long)

ML prediction time, 300 structure, 1 core  
10 min    45 min    70 min    135 min



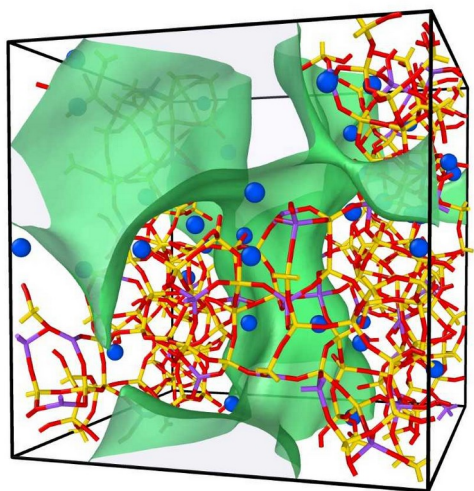
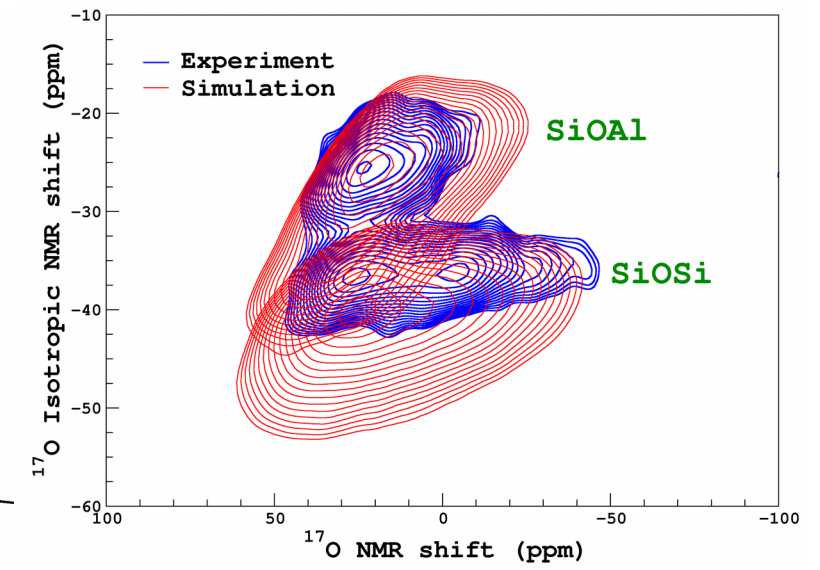
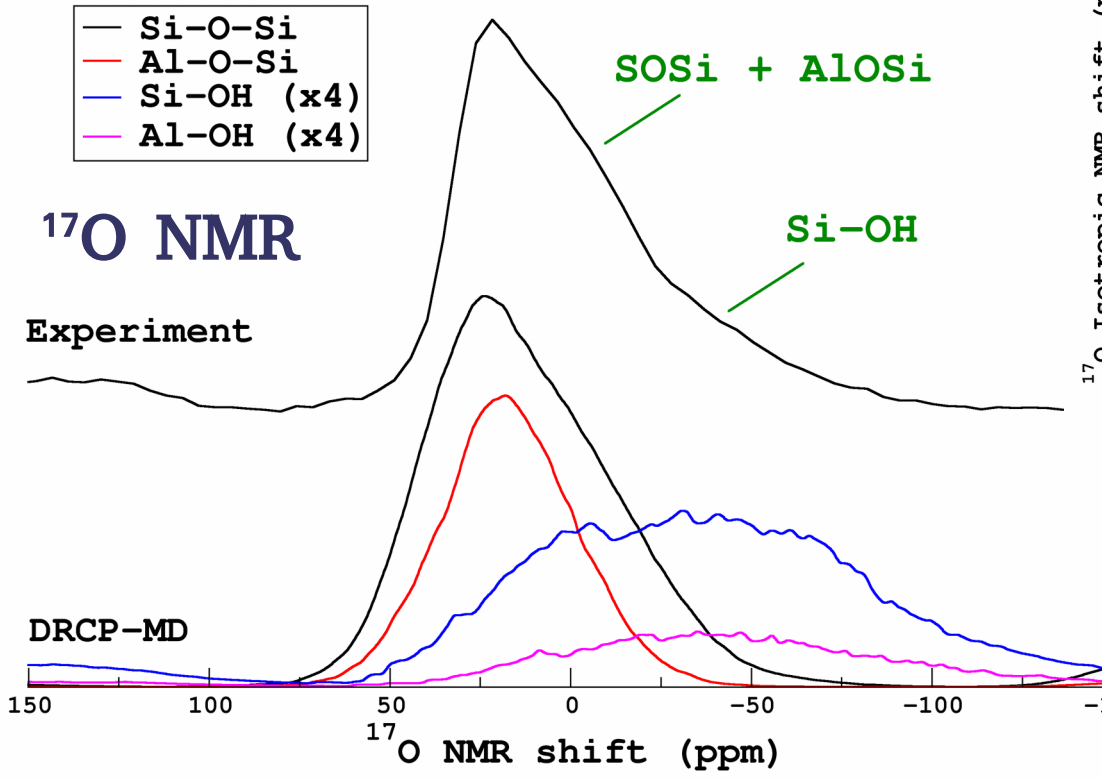
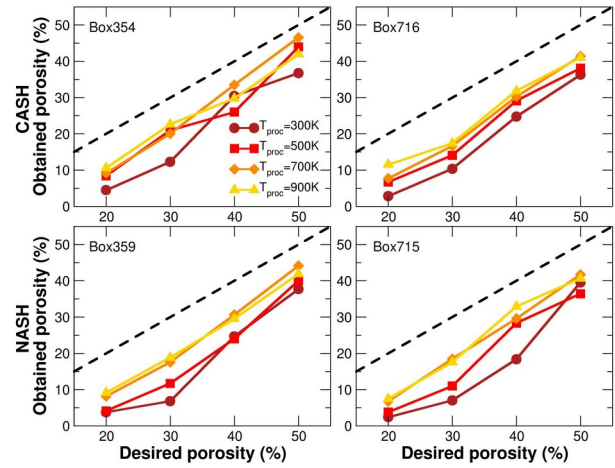
RAMAN example : Using Gaussian process regression to simulate the vibrational Raman spectra of molecular crystals, N. Raimbault, A. Grifasi, M. Ceriotti, M. Rossi September 2019 New Journal of Physics 21(10)



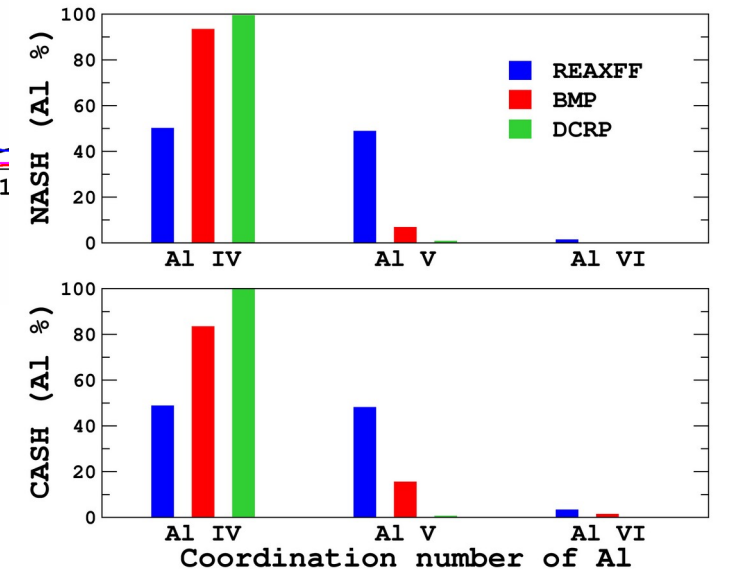
# Modelling the alteration layer

## A nanoporous hydrated aluminosilicate structures

F. Lodesani et al. In preparation.



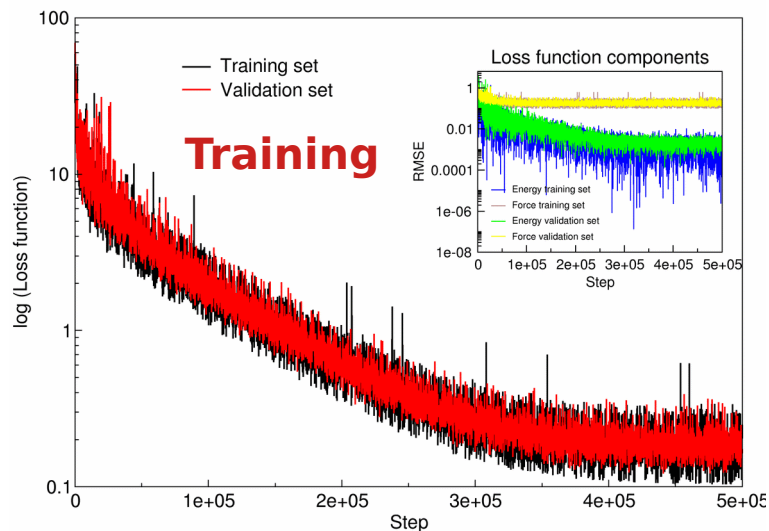
Porous structures are build with the Charge-Scaling (CS) and then filled with water molecules followed by MD (Diffuse Charge Reactive Pair potential)\* relaxation. DFT/DATABASE for ML in progress



# Machine Learning in MD : learning the descriptors

## Deep Learning in MD simulations: the next generation of force-fields ?

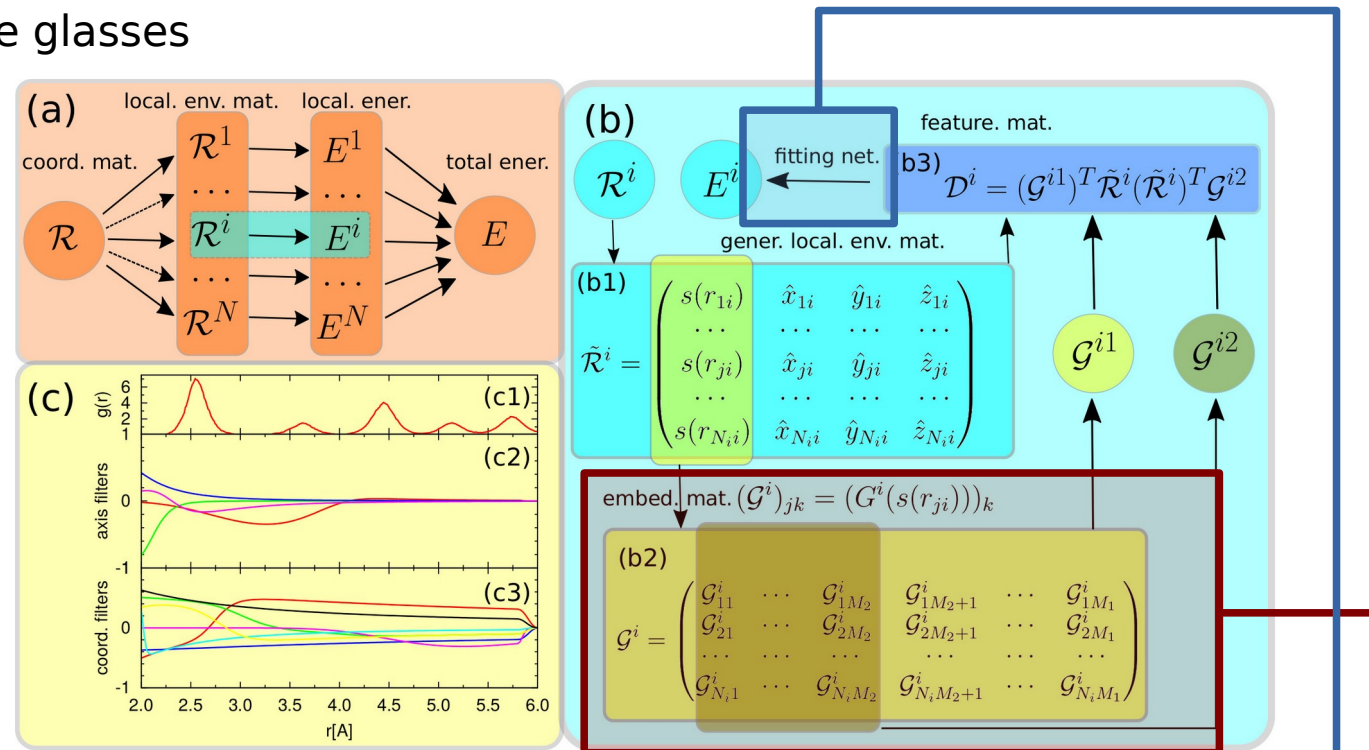
Modelling (DeepMD-SE/LAMMPS) sodium-silicate glasses



The **training** is (very) time-consuming (GPU required).

Active-Learning approach is needed: faulty environments are predicted in production mode.

They must be reintroduced in the database for a new training ...



### Deep Learning

Two Neural Networks are used:

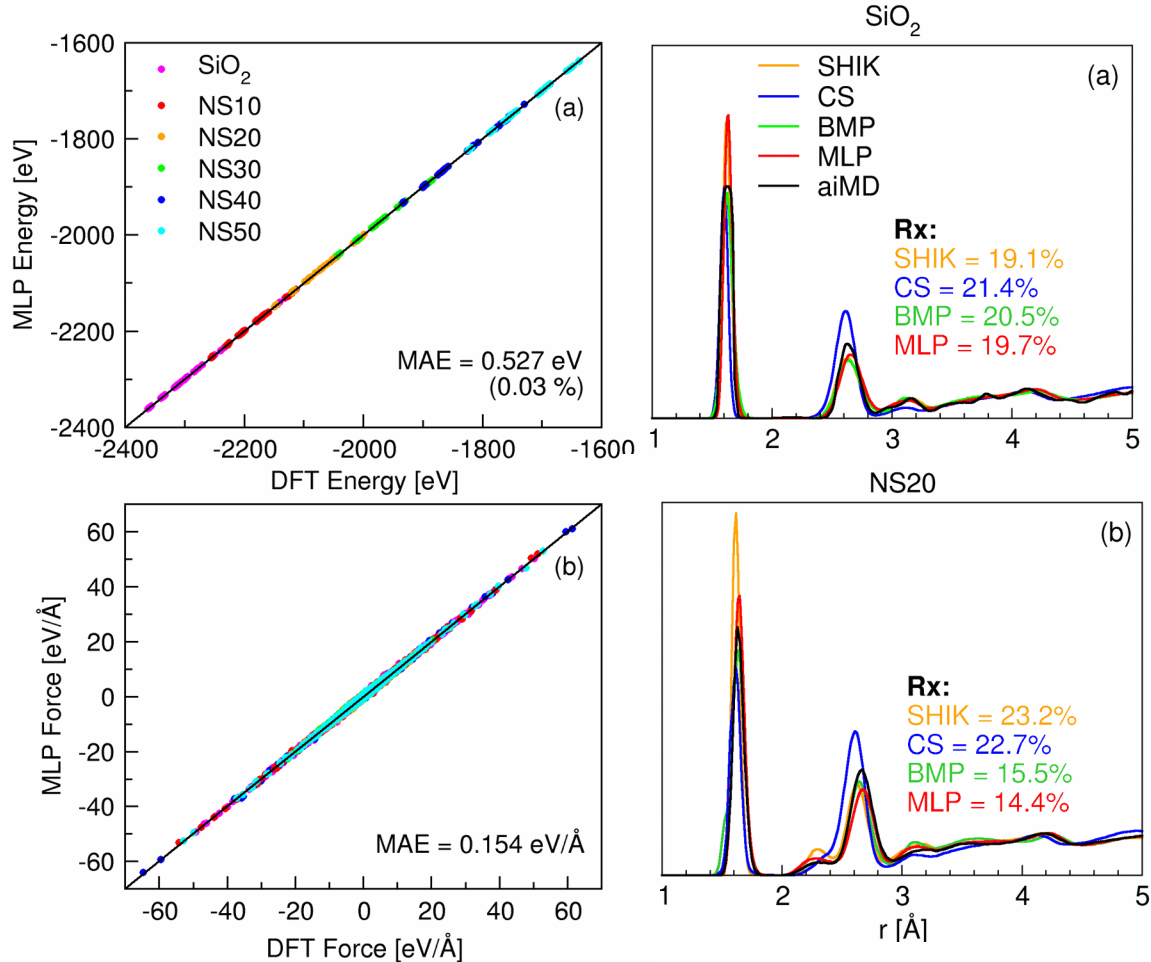
The first generates the descriptors (**embedding**)

The second performs the fitting (**regression**)

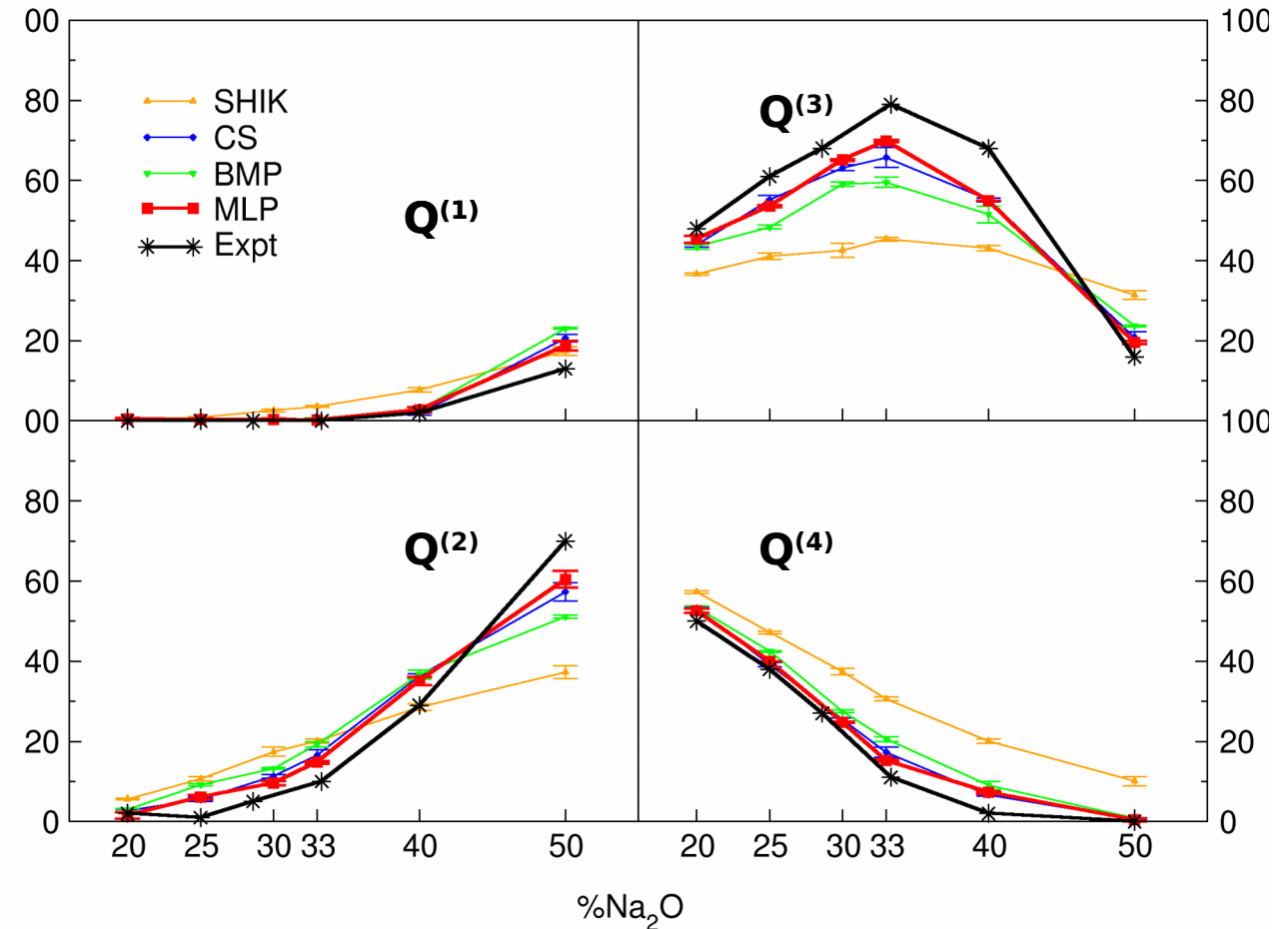
# Machine Learning in MD : learning the descriptors

## Deep Learning in MD simulations: the next generation of force-fields ?

Modelling (DeepMD-SE/LAMMPS) sodium-silicate glasses



Si Q<sup>(n)</sup> speciations



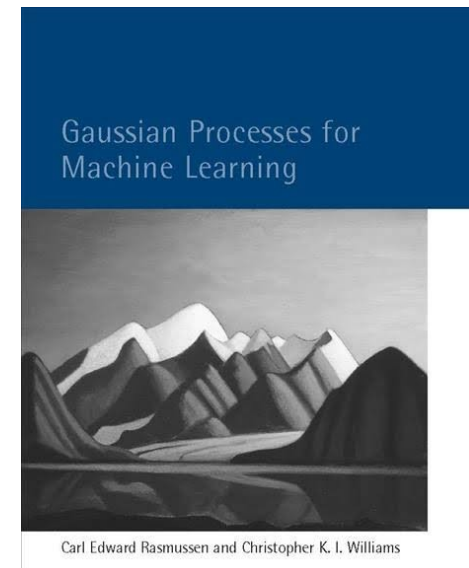
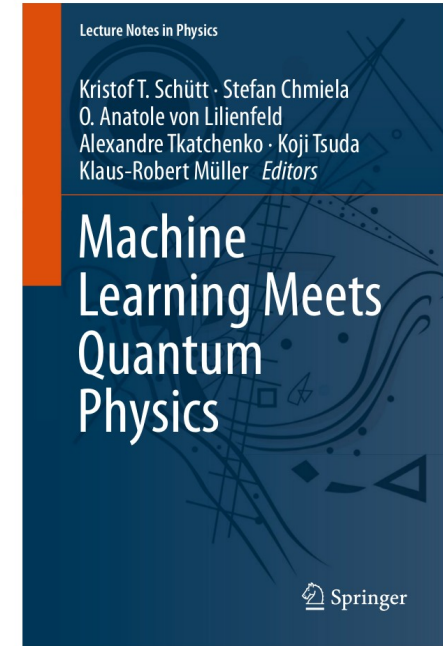
M. Bertani et al., *An Accurate and transferable Machine Learning potential for Molecular Dynamics simulation of sodium silicate glasses*, submitted.

**DeepMD-SE:** \* Zhang et al., *Deep Potential Molecular Dynamics: A Scalable Model with the Accuracy of Quantum Mechanics*. *Phys. Rev. Lett.* 120, 143001 (2018).



# Conclusions and Perspectives

- NMR provides a wealth of techniques for studying oxide glasses and their alteration (in aqueous media or in vapor phase)
- NMR experiments can be simulated from MD simulations (with DFT/GIPAW)
- MD and DFT/NMR can be accelerated with Machine Learning
- ML-MD offers exciting perspective for modelling hydrated systems (H)
- Big Database need to be constructed : aiMD/DFT simulations are needed (in progress)
- *ML vibrational fingerprint (IR, RAMAN) in progress*



# Acknowledgements



- **Frédéric Angeli, Sophie Schuller, Stéphane Gin, Sylvain Peugeot, Jean-Marc Delaye, Marie Collin, Sathya Narayanasami, Nicolas Bisbrouck**

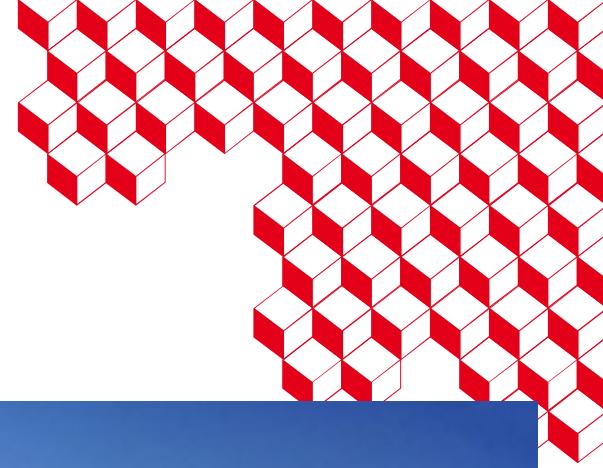
*CEA DES/ISEC Marcoule, France*

**CEA VESTALE Program**

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*UniMore, Modena, Italy*



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Thank you for your attention

**Merci de votre attention**



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